

## Microarticle

The optical bandgap of lithium niobate ( $\text{LiNbO}_3$ ) and its dependence with temperature

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## ABSTRACT

Notwithstanding the great scientific-technological interest in lithium niobate ( $\text{LiNbO}_3$ ), its optical bandgap  $E_{\text{gap}}$  has been subject of intense discussion. So far, the literature exhibits different  $E_{\text{gap}}$  values spanning over about 2 eV and comprises a mixture of compositions, structures, and theoretical methods – not always clearly indicated or discussed. In view of that, this work presents a thorough investigation of the  $E_{\text{gap}}$  (at room-temperature and in the  $\sim 80$ – $800$  K temperature range) of the congruent ferroelectric  $\text{LiNbO}_3$  (Z-cut) single crystal.

Lithium niobate ( $\text{LiNbO}_3$ ) is a manmade material that, since its advent, has fascinated the scientific community because of its non-linear, electro-optical, and photo-refractive outstanding properties [1]. As a consequence,  $\text{LiNbO}_3$  is considered the “*Silicon of Photonics*” with practical applications ranging from wavelength (or surface acoustic wave) filters to optical waveguides (or modulators) and optical frequency converters (or oscillators) – just to mention a few of them [2]. In most of these applications, crystalline  $\text{LiNbO}_3$  wafers are prepared by *Czochralski* – in which the X-, Y-, or Z-cuts indicate the crystallographic axes that are normal to the large wafer surfaces – and exhibit a non-stoichiometric composition (typically  $48.5 \pm 0.5$  mol% of  $\text{Li}_2\text{O}$ ), rendering a Li-deficient structure and lots of defects [1,3]. Contrasting with these so-called congruent  $\text{LiNbO}_3$ , better quality stoichiometric  $\text{LiNbO}_3$  can be achieved, though at higher costs. In terms of properties, it is common sense that, below its *Curie* temperature  $\Theta_{\text{Curie}}$  ( $\sim 1420$  K),  $\text{LiNbO}_3$  is ferroelectric (space group  $R\bar{3}c$ ) consisting of oxygen octahedra sharing faces along the 3-fold axis [1]. Paraelectric  $\text{LiNbO}_3$  (space group  $R\bar{3}c$ ) is stable only above  $\Theta_{\text{Curie}}$  and, therefore, has received comparatively less attention than its ferroelectric phase. Notwithstanding such scientific interest and its many successful technological achievements, the optical bandgap  $E_{\text{gap}}$  of  $\text{LiNbO}_3$  is still under debate. This seems somewhat contradictory given the importance of the  $E_{\text{gap}}$  (value and behavior) in developing practical devices but, at present, the literature regarding the  $E_{\text{gap}}$  of  $\text{LiNbO}_3$  is rather diffuse. Just to illustrate the point, it is usual to find experimental  $E_{\text{gap}}$  values in the 3.3–4.7 eV [4–[7]] range, along with those provided by theoretical figures ranging from 3.5 to 6.5 eV [8]. Whereas most of the discrepancies in the experimental  $E_{\text{gap}}$  values arise because of differences in the composition and structure of  $\text{LiNbO}_3$  (i.e.: congruent vs stoichiometric, ferroelectric vs paraelectric, pure vs doped materials, etc.), the theory behind  $E_{\text{gap}}$  seems to be highly influenced by the calculation methods – inputs as well. Another issue is

related to the nature of the optical transitions in  $\text{LiNbO}_3$  – most likely indirect [4,5] (in spite of some misleading citations [8]).

The above scenario form the basis of this work that investigates the optical bandgap  $E_{\text{gap}}$  of congruent  $\text{LiNbO}_3$ , as determined by optical transmission measurements as a function of temperature. In fact, in contrast to the only existing temperature-dependent  $E_{\text{gap}}$  report on  $\text{LiNbO}_3$  [9], this paper presents a detailed compositional – structural analysis of the  $\text{LiNbO}_3$  crystal as well as a comprehensive analysis of  $E_{\text{gap}}$  according to standard procedures.

The sample considered in this work corresponds to a commercial (congruent, ferroelectric, undoped, optical grade, Z-cut, 2-side polished)  $\text{LiNbO}_3$  single crystal. Both the atom structure and composition of the sample was verified by Raman spectroscopy (backscattering geometry, 632.8 nm excitation). Optical transmittance ( $\mathcal{T}$ ) and reflectance ( $\mathcal{R}$ ) measurements were carried out in the  $\sim 250$ – $1000$  nm range by means of a miniature spectrophotometer, optical fibers, and integrating sphere. All spectra were properly corrected by the optical response of the system (light source + diffraction grating + detector) to ensure 100% light transmission or reflection (from a Al mirror reflectance standard). Additional  $\mathcal{T}$  measurements were performed in the 83–773 K temperature range (in steps of either 25 or 50 K) with the sample placed in a temperature-stage. In this case, a dwell time of 3 min was adopted before each measurement to allow the sample to reach thermal equilibrium.

A typical Raman spectrum of the  $\text{LiNbO}_3$  single crystal, taken in the  $z(x, xy)\bar{z}$  orientation, is shown in Fig. 1(a). Exactly the same spectrum was achieved by imposing different laser and/or detection polarization conditions (and after sample rotation), confirming that the crystal corresponds to the Z-cut [10].

In addition to the crystal quality and orientation, the Raman spectrum was considered to assess the  $\text{LiNbO}_3$  composition. The estimate is based on the fact that any change in the  $\text{LiNbO}_3$  crystal lattice (i.e.:

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translational symmetry, atoms masses, and force constants) influences its internal potential. As a result, in non-stoichiometric  $\text{LiNbO}_3$ , the substitution of Li by Nb atoms modifies the overall phonon behavior increasing the linewidth (FWHM) of certain phonon modes. The concept was originally proposed by Schlarb *et al.* and, since then, it has been used to infer the relative amount of  $\text{Li}_2\text{O}$  in  $\text{LiNbO}_3$  crystals – simply by applying the experimentally determined linewidths of the  $\text{E}(\text{TO}1)$  and/or  $\text{A}1(\text{LO}4)$  modes into some calibration curves [11]. Fig. 1(b) and 1(c) show the *Lorentzian* curve fittings of these modes, according to which the average  $[\text{Li}_2\text{O}]$  was found to be  $48.5 \pm 0.3 \text{ mol\%}$ .

Fig. 1(e) shows the optical absorption coefficient  $\alpha(E)$  – as obtained from the room-temperature  $T$  and  $R$  spectra of Fig. 1(d) – along with the sigmoid-Boltzmann curve that best reproduces the  $\alpha(E)$  data. According to this procedure the optical bandgap is given by  $E_{\text{gap}}^{\text{Boltz}} = E_0^{\text{Boltz}} - n_{\text{type}}^{\text{Boltz}} \cdot \delta E$ , where  $E_0^{\text{Boltz}}$  and  $\delta E$  correspond to the central energy and slope of the sigmoid-Boltzmann function, and the empirical  $n_{\text{type}}^{\text{Boltz}}$  parameter stands for the type of optical transition or bandgap ( $n_{\text{direct}}^{\text{Boltz}} = 0.3$  and  $n_{\text{indirect}}^{\text{Boltz}} = 4.3$ ) [12]. Given its simplicity, insensitivity to measurements – analyses problems, and  $E_{\text{gap}}$  uncertainties comparable to (or below) those exhibited by other methods, the Boltzmann-related method was adopted in this work. Accordingly, an indirect  $E_{\text{gap}}^{\text{Boltz}} = 3.77 \pm 0.05 \text{ eV}$  has been achieved for congruent (Z-cut)  $\text{LiNbO}_3$ , at room-temperature.

Because of experimental restrictions, only the transmittance  $T$  of  $\text{LiNbO}_3$  was measured as a function of temperature – a few  $\alpha(E)$  spectra, as derived from these  $T$  measurements, are displayed in Fig. 1(f). As can be seen, the spectra experience a clear red-shift at increasing temperatures. This temperature-induced effect can be evaluated, for example, by representing the  $E_{\text{gap}}$  of  $\text{LiNbO}_3$  as a function of temperature  $E_{\text{gap}}(T)$  – as shown in Fig. 2. The figure also indicates the  $E_{\text{gap}}(T)$  values of Redfield & Burke [9], as well as the  $E_{\text{gap}}$  at room-temperature of some other references. In addition to the  $E_{\text{gap}}(T)$  behavior, it is impressive the data dispersion at room-temperature. According to Fig. 2, the room-temperature  $E_{\text{gap}}$  of congruent  $\text{LiNbO}_3$  occurs predominantly at  $\sim 3.8 \text{ eV}$ , even though certain  $E_{\text{gap}}$  values at  $3.3 \text{ eV}$  and  $4.7 \text{ eV}$  – this time, however, regarding materials with unknown composition [6] and/or no clear atomic structure [7]. Besides, the  $E_{\text{gap}}(T)$  of reference 9 corresponds to a mix of congruent  $\text{LiNbO}_3$  crystals of different orientations and thicknesses and, particularly, by supposing  $E_{\text{gap}}$  as the energy at which  $\alpha = 2000 \text{ cm}^{-1}$ .

Taking into account this variety of values–conditions–methods

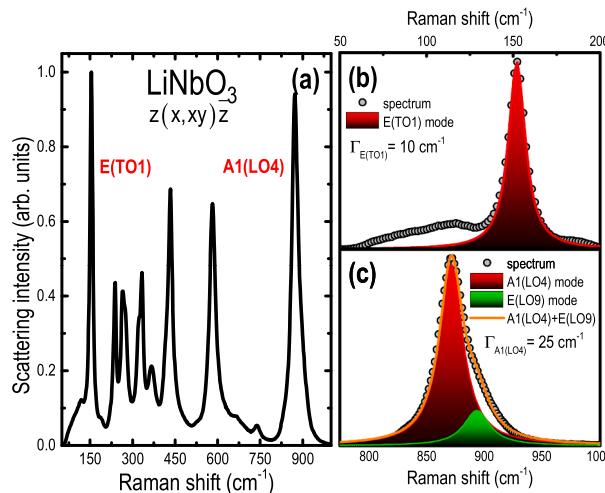


Fig. 1. (a) Room-temperature Raman scattering spectra ( $z(x,xy)\bar{z}$  orientation) of a  $\text{LiNbO}_3$  single crystal (Z-cut). (b) Detail of the  $\text{E}(\text{TO}1)$  phonon mode and its corresponding *Lorentzian* curve fitting (FWHM =  $\Gamma_{\text{E}(\text{TO}1)} = 10 \text{ cm}^{-1}$ ). (c) Detail of the  $\text{A}1(\text{LO}4)$  and  $\text{E}(\text{LO}9)$  phonon modes and corresponding curve fittings ( $\Gamma_{\text{A}1(\text{LO}4)} = 25 \text{ cm}^{-1}$ ). (d) Optical transmittance  $T$  and reflectance  $R$  spectra of  $\text{LiNbO}_3$ . (e) Respective absorption coefficient  $\alpha$ , at room-temperature, indicating an indirect  $E_{\text{gap}}^{\text{Boltz}} = 3.77 \text{ eV}$ . (f) Absorption coefficient spectra (as obtained from  $T$ ) at various temperatures.

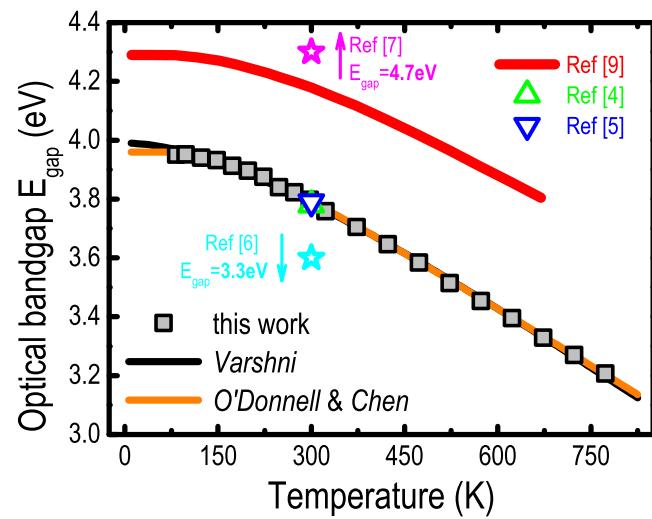
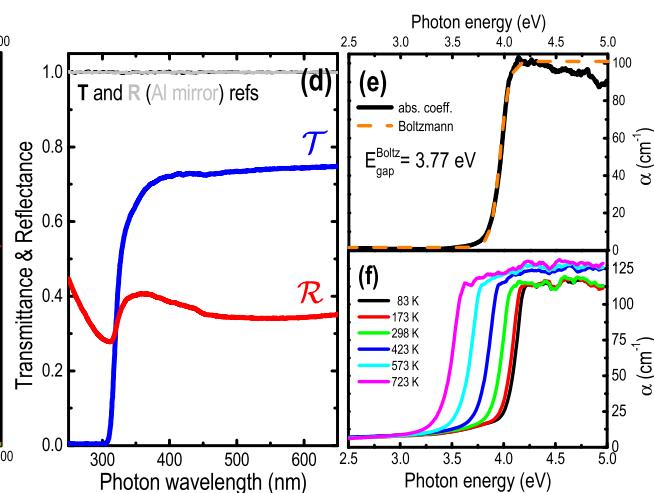


Fig. 2. Optical bandgap  $E_{\text{gap}}$  of a  $\text{LiNbO}_3$  crystal (congruent, Z-cut) as a function of temperature – as obtained in this work, along with some theoretical expressions (by Varshni [13] and O'Donnell & Chen [15]) – as well as some data from literature.

from literature, the experimental results obtained in the present work (based on a well-characterized crystal and following standard procedures) were considered to explore the temperature-dependent  $E_{\text{gap}}$  behavior of  $\text{LiNbO}_3$ . In fact, both  $E_{\text{gap}}$  and  $E_{\text{gap}}(T)$  are fingerprints of many materials and knowing their values and behavior are essential to produce new-improved semiconductor-related devices. Traditionally, the study of  $E_{\text{gap}}(T)$  involves the analysis of the experimental data according to the empirical description by Varshni [13], such that  $E_{\text{gap}}(T) = E_{\text{gap}}(0) - \frac{P_1 T^2}{P_2 + T^2}$  where  $E_{\text{gap}}(0)$  stands for the  $E_{\text{gap}}$  at  $T = 0 \text{ K}$ , and  $P_1$  and  $P_2$  are fitting parameters (characteristic of each material). Further studies consider the separation of  $E_{\text{gap}}(T)$  into contributions: due to lattice expansion/contraction effects and electron–phonon interactions [14]. Although most of these approaches reproduce the experimental  $E_{\text{gap}}(T)$  with some precision, unfortunately, not all of them are straightforward and/or provide realistic physical information. An alternative way to overcome part of these problems – still involving an empirical treatment, but based on simple thermodynamic concepts – was proposed by O'Donnell & Chen [15]. Roughly, the model takes into consideration the



similar (average) temperature-dependent effect that crystal lattice and electron–phonon coupling have on  $E_{\text{gap}}$ , according to which the experimental data can be described by  $E_{\text{gap}}(T) = E_{\text{gap}}(0) - S\langle\hbar\omega\rangle\left[\coth\left(\frac{\langle\hbar\omega\rangle}{2k_B T}\right) - 1\right]$ , where  $E_{\text{gap}}(0)$  is the bandgap value at  $T = 0$  K,  $S$  is a dimensionless coupling constant, and  $\langle\hbar\omega\rangle$  is an average phonon energy. The theoretical analysis of the present  $E_{\text{gap}}(T)$  data is shown in Fig. 2, according to which it is clear the differences between the models by *Varshni* and *O'Donnell & Chen* – in special at very low temperatures. Furthermore, the approach by *O'Donnell & Chen* perfectly fits the experimental  $E_{\text{gap}}(T)$  data in the whole 83–773 K temperature range providing quite reasonable values with  $\hbar\omega \sim 40$  meV (*i.e.*, on the order of the typical phonon frequencies of  $\text{LiNbO}_3$  – see Fig. 1(a)), and presenting  $E_{\text{gap}}$  values somehow stationary below  $T \sim 50$  K.

In summary, this work presents the first comprehensive study of the optical bandgap  $E_{\text{gap}}$  of the  $\text{LiNbO}_3$  crystal (congruent, Z-cut). Accordingly, the indirect optical bandgap of  $\text{LiNbO}_3$  was found to comply with  $E_{\text{gap}}^{\text{IND}}(T) = 3.96 - 0.311\left[\coth\left(\frac{0.04}{2k_B T}\right) - 1\right]$ , rendering  $E_{\text{gap}}^{\text{IND}}(T) = 3.77 \pm 0.05$  eV at room-temperature.

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## Declaration of Competing Interest

The authors declare that they have no known competing financial interests or personal relationships that could have appeared to influence the work reported in this paper.

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