



## Amino acid decorated xanthan gum coatings: Molecular arrangement and cell adhesion

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### ABSTRACT

In this work, citric acid molecules mediated simultaneously the crosslinking of xanthan gum (XG) chains and the grafting of glutamic acid (Glu), cysteine (Cys), histidine (His) or tryptophan (Trp) to XG chains, as evidenced by Fourier-transform infrared spectroscopy, elemental and thermogravimetric analyses, and high-resolution X-ray photoelectron spectroscopy. XG coatings (~60 nm thick) presented total surface energy ( $\gamma_S$ ) of 65 mJ/m<sup>2</sup>. The attachment of amino acids to the XG chains decreased  $\gamma_S$  values to ~48 mJ/m<sup>2</sup> because the surface energy polar ( $\gamma_p$ ) component decreased. Circular dichroism (CD) spectra revealed that the helix conformation of XG chains was retained in XG and XG-Glu-coatings, but it was lost in the XG-His, XG-Cys and XG-Trp-coatings due to intermolecular interactions between the amino acid side groups (thiol, imidazole, and indole). XG-based coatings presented no cytotoxicity. After 3 h of incubation, the adhesion of SH-5YSY neural cells in comparison to the control (100%) was XG-His (18%), XG-Cys (21.4%), XG (29%), XG-Trp (28.6%) and XG-Glu (46.4%). The decrease of % cell adhesion tended to be favored by the decrease of  $\gamma_p$ , except for XG-Glu and XG-Trp, and by the loss of helix molecular conformation of chains in the coatings.

### 1. Introduction

Polysaccharides have been frequently applied in the development of biomedical devices due to their biocompatibility, water absorption, biodegradability, renewable sources, and the possibility to conjugate with others molecules (Bas, Catelas, De-Juan-Pardo & Huttmacher, 2018).

Cell adhesion might be undesirable for specific applications, such as for heart valves devices such as stents, implants, or even injection syringes because the cells are able to adhere and grow on the blood channel, blocking the blood flow (Khalili & Ahmad, 2015). Neural devices such as neural electrodes are a “hot topic” due to their capability of cell-stimulating and signal recording in the brain Yin, Liu, Xiao and Zhang (2021). However, the devices can fail due to fouling caused by astrocytes and microglial cells surrounding the electrode, promoting inflammation processes and scar formation (Polikov, Tresco & Reichert, 2005). This drawback can be overcome by the addition of anti-fouling coatings on the electrodes. For instance, in comparison to the

uncoated neural silicon and iridium electrodes, microglia and astrocytes attached less after coating with poly(*N*-isopropyl acrylamide)-co-poly (acrylic acid) crosslinked with poly(ethylene glycol) (PEG) (Gutowski et al., 2014). Neural cell adhesion protein (L1) conjugated with PEG layer contributed to decreasing the astrocyte's attachment on silicon surfaces (Azemi, Stauffer, Gostock, Lagenaur & Cui, 2008). However, polysaccharide-based coatings designed for reducing cell adhesion are scarcely reported in the literature (F. Ceysens et al., 2013; D.H. Kim, Wiler, Anderson, Kipke & Martin, 2010).

Xanthan gum (XG) produced by the bacteria genus *Xanthomonas* is a versatile polysaccharide, which has been applied in a wide range of technological fields (Petri, 2015). Fig. 1 represents the chemical structure of XG. The cellobiose repeating unit composes the backbone and the trisaccharide D-mannose ( $\beta$ -1,4) D-glucuronic acid ( $\beta$ -1,2) and D-mannose, which are attached to alternate glucose residues in the backbone by  $\alpha$ -1,3 linkages, composes the side chain. D-mannose unit linked to the main chain contains an acetyl group at position O-6 and approximately one-half of the terminal D-mannose contains a pyruvic

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acid residue (Petri, 2015). At pH > 4, XG chains behave as polyanions due to the carboxylic acids deprotonation. XG has been applied for drug release (Bueno, Hilamatu, Carmona-Ribeiro & Petri, 2018), cartilage (Byram et al., 2020), probiotic co-encapsulation (Xiao et al., 2020), bowel healing (Huang et al., 2021), and scaffolds for neuronal differentiation (T. Glaser, Bueno, Cornejo, Petri & Ulrich, 2015; S. Konda-veeti, Semeano, Cornejo, Ulrich & Petri, 2018) due to its high biocompatibility. On the other hand, amino acids (AA) are interesting candidates for the decoration of thin films because they are fundamental organic molecules used for cell signaling, gene expression, proteins phosphorylation balance, and other biological molecules synthesis (e.g. basic unit to protein chain) (Wu, 2009). Glutamic acid (Glu) (Fig. 1), an anionic AA at pH 7.4, is responsible for immune function. Histidine (His) (Fig. 1), a cationic AA at pH 7.4, presents antioxidant and wound healing properties. Cysteine (Cys) (Fig. 1), a polar AA at pH 7.4, is an abundant source of sulfur (-SH thiol group), and tryptophan (Trp) (Fig. 1) is a hydrophobic AA with indole group.

In this work, XG coatings were individually crosslinked and decorated with Trp, His, Glu-and Cys-simultaneously on glass slides and Si wafers, with the aid of citric acid. Si wafers are generally used as substrates for neural electrodes and microelectronic devices. Citric acid (CA) is a low cost nontoxic crosslinker for polysaccharides (A.C. Alavarse et al., 2022; A.K. Antosik, Piątek & Wilpiszewska, 2019; M.A. Hussain, Kiran, Haseeb, Hussain & Hussain, 2020; Z. Qin, Jia, Liu, Kong & Wang, 2020; H. H. Wu et al., 2019). The hypothesis of this study is that the type of AA grafted to the XG coatings might affect the molecular conformation of XG chains confined in the coatings, coating surface energy and adhesion of cells on them.

## 2. Experimental

### 2.1. Materials

Xanthan gum (XG) (CP Kelco, Brazil) with degree of pyruvate of 0.39, degree of acetyl of 0.42, and  $M_v$  of  $1.3 \times 10^6$  g/mol (Supplementary Material SM1) was used as received. L-cysteine (Cys), L-glutamic acid (Glu), L-histidine (His), L-tryptophan (Trp) amino acids were provided by Sigma Aldrich (Brazil); citric acid (CA) and sodium

hypophosphite (SHP), hydrochloric acid (HCl) were purchased from Labsynth (Brazil). Silicon (100) wafers with native SiO<sub>2</sub> layer (University Wafer, USA) and glass coverslip (KASVI, Brazil) were cleaned with isopropanol prior to use.

### 2.2. Coating preparation

Solutions of 0.2 g/L XG, 0.2 g/L Cys, Glu, His-and Trp-were prepared separately in 0.01 mol/L HCl. The acid medium favors the crosslinking reaction and the solubility of AA. The crosslinking solution was composed by  $1 \times 10^{-2}$  g/L CA and  $5 \times 10^{-3}$  g/L SHP in a volume proportion to 4:1. XG, AA and crosslinking solutions were mixed in a volume proportion of 2:2:1 (XG:AA:CA/SHP) or 2:1 for XG:CA/SHP and homogenized for 0.5 h. The precursor solutions covered the freshly rinsed Si wafers ( $\sim 1$  cm<sup>2</sup>), which were kept overnight in an oven at 40 °C until complete evaporation of water. From this point on, ultrathin and thin coatings were prepared independently (Scheme 1). In order to obtain ultrathin films, a set of dry coatings was extensively rinsed in MilliQ, so that only XG chains and AA molecules closely bound to the silanol groups on the Si wafers or glass slides remained attached to the surface. Then, the samples were heated at 165 °C for 7 min in order to promote the crosslinking reaction and the grafting of AA to XG chains. Dry coatings or micrometric films prepared by casting were heated at 165 °C during 7 min. All coatings and micrometric films were rinsed in MilliQ water to remove reactants that were just physically bound. They were coded as XG (no addition of AA), XG-Cys, XG-Glu, XG-His-and XG-Trp.

### 2.3. Physicochemical characterization of XG based coatings and micrometric films

The mean thickness (d) values of ultrathin and thin coatings were determined by means of ellipsometry. Contact angle measurements were performed by the sessile drop method. The surface energy ( $\gamma_s$ ) of the XG based thin coatings was assessed by means of contact angles ( $\theta$ ) performed with droplets (10  $\mu$ L) of diiodomethane and water. The chemical composition on the thin coating surface was determined by X-ray photoelectron spectroscopy (XPS). Circular dichroism (CD)

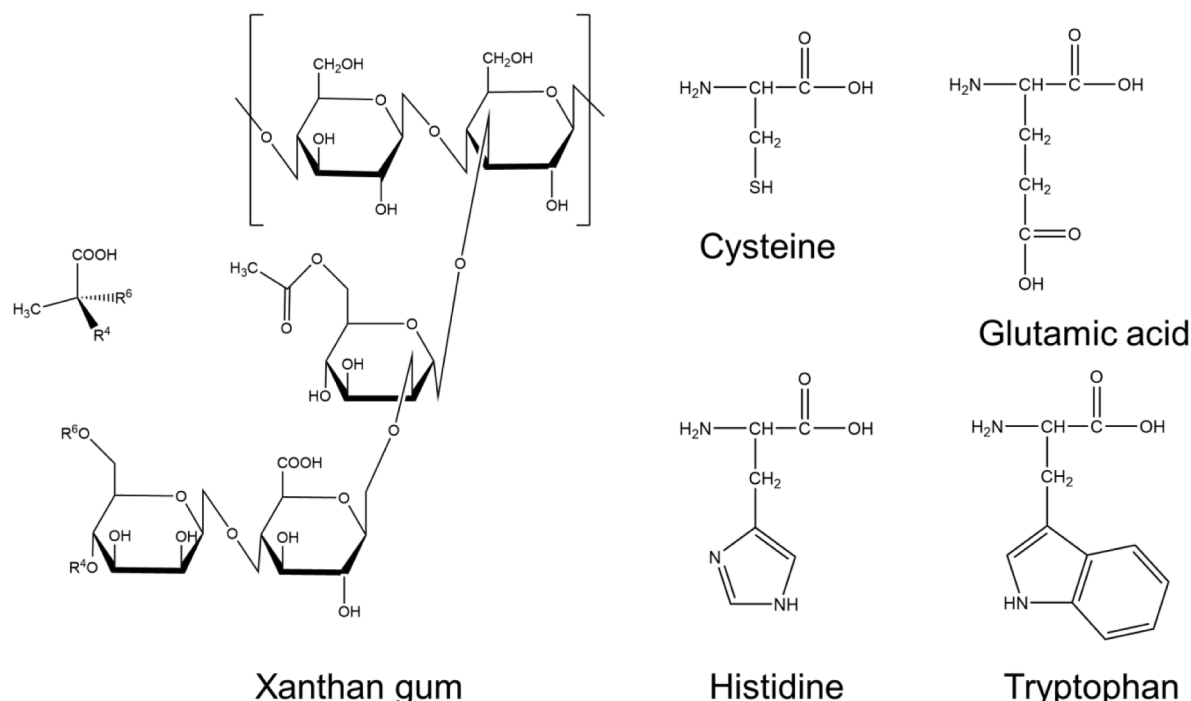
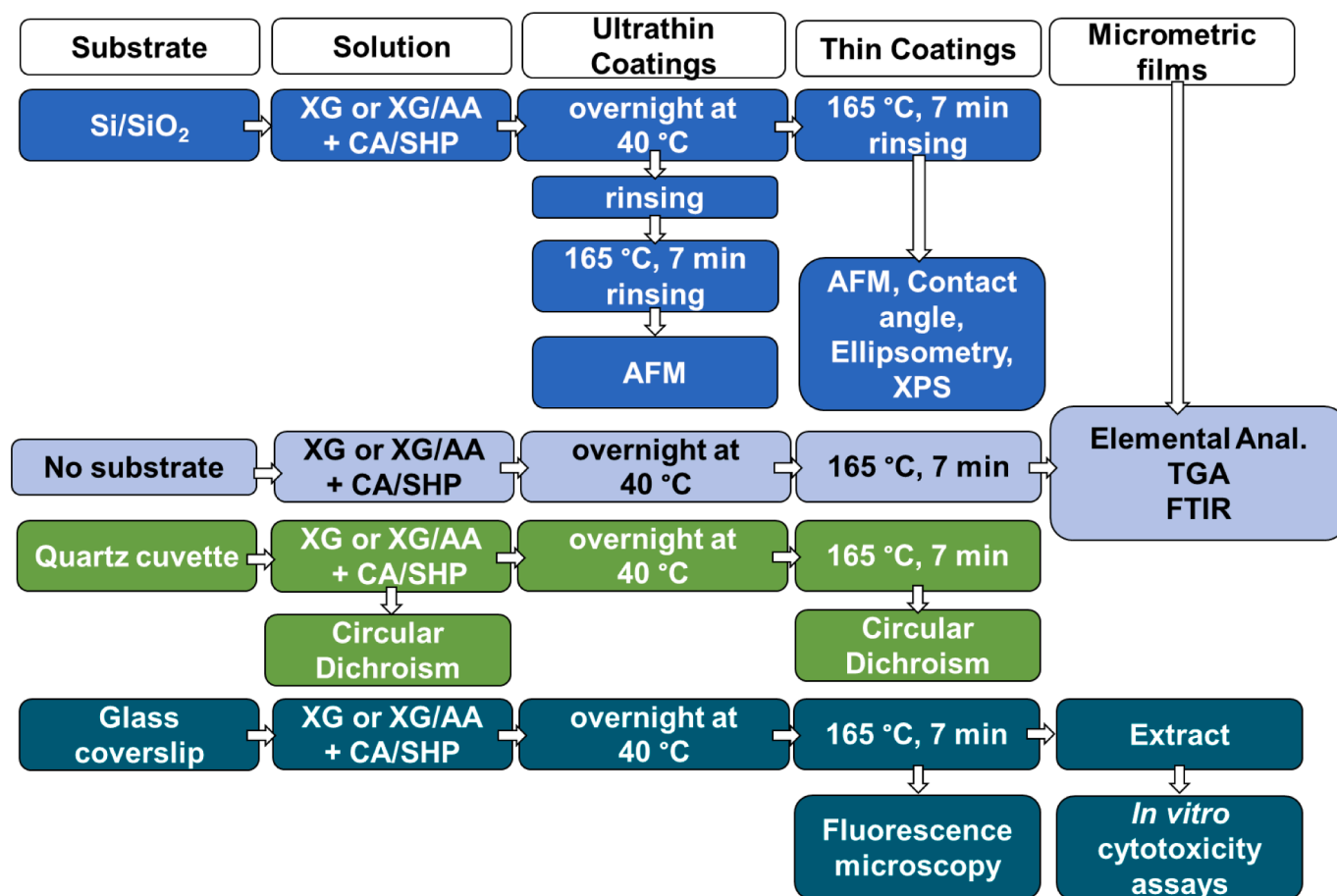


Fig. 1. Representation of chemical structures of xanthan repeated unit, cysteine (Cys), glutamic acid (Glu), histidine (His) and tryptophan (Trp).



**Scheme 1.** Overview of experimental steps for the synthesis and characterization of XG-based coatings and films for the assays with cells.

measurements were performed on thin coatings deposited on the external wall of quartz cuvettes. Topography and surface roughness of ultrathin and thin films were analyzed by atomic force microscopy (AFM). XG (powder), Cys, Glu, His, Trp and citric acid and XG, XG-Cys, XG-Glu, XG-His and XG-Trp-micrometric ( $\sim 50 \mu\text{m}$  thick) films were analyzed by Fourier-transform infrared vibrational spectroscopy in the attenuated total reflectance mode (FTIR-ATR). The amount of AA grafted to the XG chains was quantified by elemental analysis (CNH). The thermal behavior of XG, XG-Cys, XG-Glu, XG-His and XG-Trp-micrometric ( $\sim 50 \mu\text{m}$  thick) films and XG (powder) was investigated by thermogravimetric analyses (TGA). Details about equipment used and measurement conditions were provided as **Supplementary Material SM2**.

#### 2.4. Cell adhesion and viability

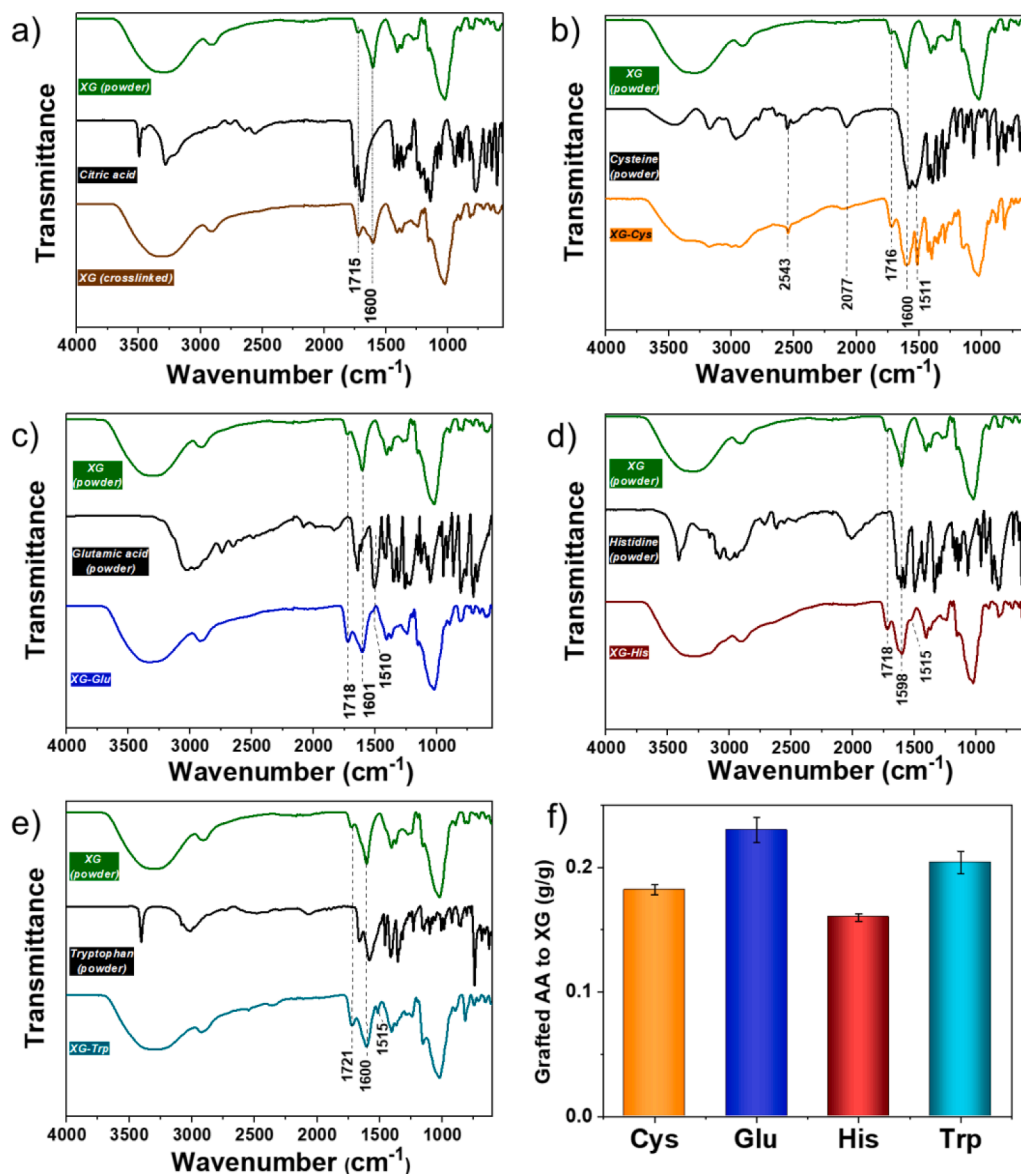
SH-SY5Y neuroblastomas cells (ATCC, CRL-2266) cloned from human bone marrow tissue were used for the cell adhesion viability assays. The in vitro cytotoxicity assays were performed using the solution extract, according to H. ISO 10993-5:2009 (H. ISO 10993-5:2009 - Biological Evaluation of Medical Device), which involves the contact between cells seeded into the well plates with extract solution derived from the thin coatings. Metabolic activity was evaluated by the 3-(4,5-dimethyl-2-thiazolyl)-2,5-diphenyl-2H-tetrazolium bromide (MTT) assay. The cell viability of each sample was tested in quintuplicate ( $n = 5$ ), dividing the absorbance reading corresponding to the sample by the absorbance reading of the control. After running the MTT assays, the remaining solution from these tests was used to quantify the lactate dehydrogenase (LDH) activity with the aid of LDH Liquiform kit assay (Labtest, Brazil). In order to observe the cells' adhesion on the thin

coatings, SH-SY5Y cells ( $5 \times 10^5$  cells/well) were cultivated for 3 h and 24 h. After that, cells were fixed with paraformaldehyde solution (4%) for 30 min at  $4^\circ\text{C}$ . The DNA of cell nuclei was stained with DAPI and the fluorescence microscopy images were recorded using an inverted fluorescence microscopy (Axiovert ZEISS, Germany) with an exposure time of 20 ms. The images were treated with ImageJ software (cell counter notice plug-in). The total cell count took into account the acquisition of three random images for each coating. Details about equipment used and measurement conditions were provided as **Supplementary Material SM3**.

### 3. Results and discussion

#### 3.1. Crosslinking of xanthan gum chains and grafting of amino acids to them

The mechanism of crosslinking polysaccharides with polycarboxylic acids catalyzed by SHP involves the formation of anhydrides, and subsequent nucleophilic attack of polysaccharide hydroxyl groups to the anhydride, yielding ester linkages (A.C. Alavarse et al., 2022; V.S. Ghorpade, Dias, Mali & Mulla, 2019; B. Ji, Tang, Yan & Sun, 2015). Fig. 2a shows the FTIR spectra obtained for XG (powder), CA (powder) and XG crosslinked micrometric film. The spectra of XG powder and XG crosslinked film show the characteristic bands of polysaccharides in the  $3500\text{--}3200 \text{ cm}^{-1}$  region (OH vibrational stretching); at  $2930 \text{ cm}^{-1}$  and  $2850 \text{ cm}^{-1}$  (symmetrical and asymmetrical CH stretching); and in the  $1240\text{--}850 \text{ cm}^{-1}$  region (C-O and C-C stretching vibrations of the saccharide ring) (Silverstein, Bassler & Morrill, 1991). Particularly important are the intensities ratio between carbonyl stretch (C=O) bands at  $1600 \text{ cm}^{-1}$  and  $1719 \text{ cm}^{-1}$ , which correspond to the acidic and



**Fig. 2.** FTIR-ATR spectra obtained for XG (powder), CA (powder), AA (powder), XG and XG-AA crosslinked micrometric films: (a) no AA, (b) AA = Cys, (c) AA = Glu, (d) AA = His-and (e) AA = Trp. (f) Amount of grafted AA (in gram) per gram of XG-AA film, calculated from CHN analyses (**Supplementary Material SM7**).

ester forms, respectively,  $I_{1600}/I_{1719}$ . For XG powder and XG films they amounted to 3.5 and 1.2 (**Supplementary Material SM4**), respectively, indicating the increase of ester linkages after crosslinking; a similar trend was observed in previous study about the crosslink of XG with citric acid (Bueno, Bentini, Catalani & Petri, 2013).

Figs. 2b to 2e show the FTIR-ATR spectra obtained for XG (powder), AA (powder) and XG-AA crosslinked micrometric films. In general, the spectra of XG-AA showed the characteristic bands of XG and the corresponding AA, as evidenced by the band assignments in the **Supplementary Material SM5**. The XG-Cys-spectrum (Fig. 2b) presented the carbonyl stretch (C = O) bands at 1600 cm<sup>-1</sup> and 1716 cm<sup>-1</sup> corresponding to the acidic and ester forms and typical bands of Cys-thiol and amine groups at 2543 cm<sup>-1</sup> (S-H), 2077 cm<sup>-1</sup> (stretching vibrations of -NH<sub>3</sub><sup>+</sup> group of L-cysteine) and 1511 cm<sup>-1</sup> (bending N-H). The spectra of XG-Glu, XG-His and XG-Trp in Figs. 2c, 2d and 2e, respectively, presented the carbonyl stretch (C = O) bands at 1600 cm<sup>-1</sup> and ~ 1718 cm<sup>-1</sup> corresponding to the acidic and ester forms and at 1511 cm<sup>-1</sup>, that was assigned to the bending N-H (Silverstein et al., 1991).

FTIR-ATR spectra provided qualitative evidences for the crosslinking and grafting of AA onto XG chains mediated by CA molecules. CA

molecules attached to the XG chains can be converted to anhydride again, reacting with XG hydroxyl groups (crosslinking between XG chains, ester formation) or with AA amine groups (grafting of AA to the XG chains, amide formation) (Toledo, Bernardinelli, Sabadini & Petri, 2020), as depicted in the **Supplementary Material SM6**. In order to quantify the amount of AA grafted to the XG chains, elemental analyses (CHN) were performed for micrometric XG and XG-AA films (**Supplementary Material SM7**). Fig. 2f shows that in average the amount of grafted AA to the XG chains was ~ 0.20 g/g.

The effect of XG chains crosslinking and grafting of AA onto XG chains mediated by CA on their thermal stability was investigated by means of TGA (**Supplementary Material SM8**). All samples presented loss of water (8.9–15.5%) at temperatures lower than 100 °C. XG (powder) presented an event at 295 °C attributed to chain degradation (53.3% of mass loss); this event was also observed for XG and XG-AA films at similar temperature. After crosslinking XG chains with citric acid, a mass loss of 9.6% at 227 °C was attributed to the degradation of citric acid (Barbooti & Al-Sammerrai, 1986). XG-Cys, XG-Glu and XG-His also presented a mass loss at ~ 227 °C, whereas for XG-Trp this event was at ~ 211 °C. The decomposition temperatures of pure Cys,

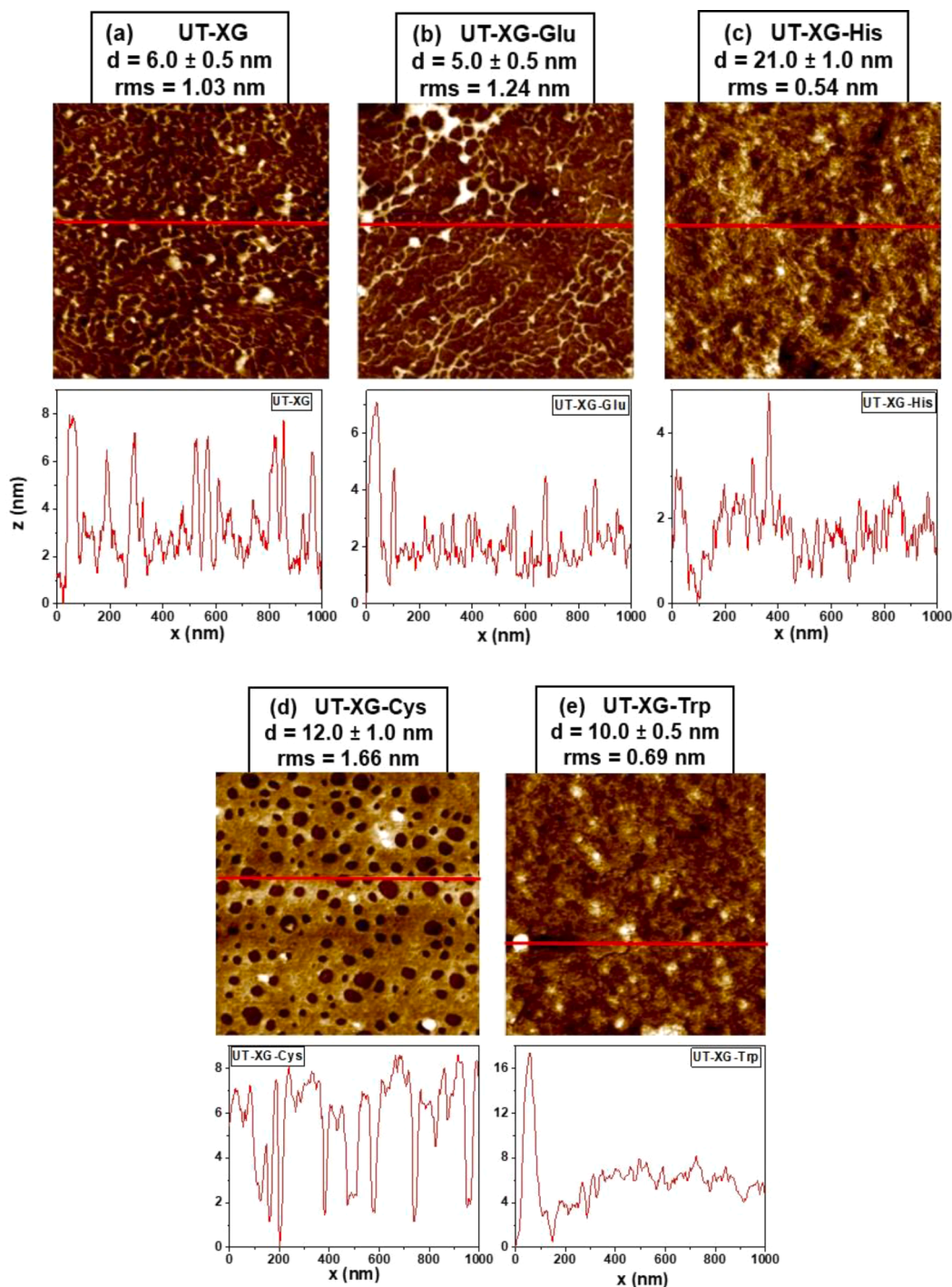


Glu-and His-are reported as 201 °C, 245 °C and 280 °C, respectively, with CO<sub>2</sub>, NH<sub>3</sub> and H<sub>2</sub>O as typical decomposition products (Weiss, Muth, Drumm & Kirchner, 2018). Under N<sub>2</sub> atmosphere pure Trp-presents two decomposition peaks, one at 304 °C and 415 °C, the latter results from the decomposition of the indole group (Da et al., 2015). XG-Cys, XG-Glu-and XG-His-films presented mass loss at ~ 200 °C that was attributed to the decomposition of amino acids. In the case of XG-Trp, the two decomposition peaks appeared at 272 °C and 362 °C. Except for Glu, grafting the AA to the XG chain decreased considerably their thermal stability.

### 3.2. Ultrathin XG based coatings

Fig. 3 shows AFM images (1 µm x 1 µm) of ultrathin (UT) coatings along with the corresponding roughness (rms) and mean thickness (d) values. The structures represent the arrangement of XG chains in the absence or presence of AA close to the Si/SiO<sub>2</sub> surface, which remained after extensively rinsing (Scheme 1). The isoelectric point of SiO<sub>2</sub> is reported as 2.8 (Chevalier et al., 2016). Thus at pH 2.0, most silanol groups tend to be positively charged and ion-dipole interactions with the XG and AA-Glu-chains are expected.

UT-XG (Fig. 3a) and UT-XG-Glu (Fig. 3b) were similar (d ~ 5–6 nm);



**Fig. 3.** AFM topographic images of ultra-thin coatings: UT-XG (a), UT-XG-Cys (b), UT-XG-Glu (c), UT-XG-His (d) and UT-XG-Trp (e). The red line represents the position corresponding to the cross-sections.

in both cases XG chains appear as crosslinked fibers and did not covered completely the underlying substrate. Similar fibrillar structures were observed by AFM for XG sample in air or liquid conditions (Liang et al., 2014) or in the presence of  $\text{Ca}^{2+}$  ions (Dário, Hortêncio, Sierakowski, Neto & Petri, 2011). The high molecular weight of XG chains and the transition from random coils conformation to ordered helix conformation driven by intermolecular interactions favor this kind of structural organization as fibers (Morris, Rees, Young, Walkinshaw & Darke, 1977). At pH 2, the carboxylic acid groups belonging to XG and Glu-are protonated, favoring intermolecular H bonds.

The presence of a positively charged side chain in His-led to the formation of smooth and homogeneous UT-XG-His-coatings (Fig. 3c); the attachment of His-groups to XG chains might favor electrostatic interactions with neighbor chains and substrate, yielding thick ( $d \sim 20$  nm) and uniform layers. The low polarity of thiol side group might have induced the formation of pores in UT-XG-Cys-coatings (Fig. 3d) during the drying process, resembling dewetting. On the other hand, the presence of hydrophobic indole group led to smooth UT-XG-Trp-coatings  $\sim 10$  nm thick (Fig. 3e); the Trp-molecules grafted to the XG chains might interact with other by hydrophobic interactions among the indole groups, orientating to the air (hydrophobic medium). These findings indicated that the type of amino acid side group induced different interactions among the AA decorated XG chains and substrate, influencing the coating morphology.

### 3.3. Thin XG based coatings

The effect of AA type grafted to the XG chains on the morphology of thin ( $d > 50$  nm) coatings was evaluated. Fig. 4 shows that networks of fibers or nanopores were not observed on thin coatings. Regardless of the AA type the surfaces presented similar features like rms and mean thickness values (Table 1). These findings indicate that the type of amino acid side group induced different interactions among the AA decorated XG chains only when the chains were confined as ultrathin films. Upon

**Table 1**

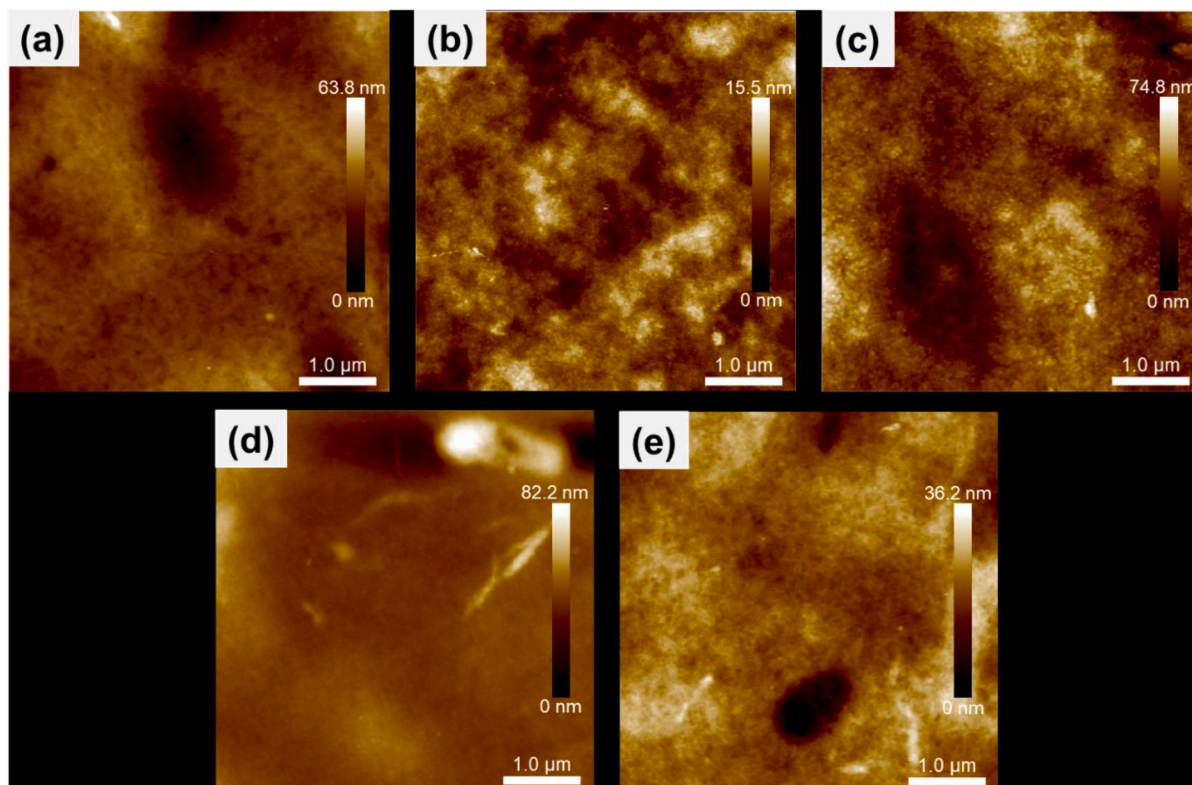
Mean thickness ( $d$ ) and roughness ( $\text{rms}$ ) values determined for thin coatings ( $n = 3$ ). Mean contact angle ( $\theta$ ) values determined for MilliQ water and  $\text{CH}_2\text{I}_2$  droplets on XG based coatings, along with the polar ( $\gamma_p\text{S}$ ) and dispersive ( $\gamma_d\text{S}$ ) components of the surface energy ( $\gamma\text{S}$ ) ( $n = 4$ ).

Sample	$d$ (nm)	$\text{rms}$ (nm)	$\theta_a$ ( $^\circ$ ) water	$\theta_a$ ( $^\circ$ ) $\text{CH}_2\text{I}_2$	$\gamma_p\text{S}$ (mJ/ $\text{m}^2$ )	$\gamma_d\text{S}$ (mJ/ $\text{m}^2$ )	$\gamma\text{S}$ (mJ/ $\text{m}^2$ )
XG	$64.8 \pm 5.6$	$12.0 \pm 2.6$	$39 \pm 4$	$40.2 \pm 0.1$	$25 \pm 2$	$40 \pm 4$	$65 \pm 6$
XG-Cys	$65 \pm 15$	$15.7 \pm 5.1$	$52 \pm 5$	$39.5 \pm 0.5$	$17 \pm 2$	$40 \pm 4$	$57 \pm 6$
XG-Glu	$66.8 \pm 9.5$	$23.0 \pm 1.6$	$63 \pm 6$	$46 \pm 2$	$12 \pm 1$	$37 \pm 4$	$49 \pm 5$
XG-His	$57 \pm 14$	$23.6 \pm 1.9$	$65 \pm 5$	$47 \pm 2$	$11 \pm 1$	$36 \pm 4$	$47 \pm 5$
XG-Trp	$62 \pm 12$	$12.0 \pm 2.5$	$63 \pm 5$	$47 \pm 2$	$12 \pm 1$	$36 \pm 4$	$48 \pm 5$

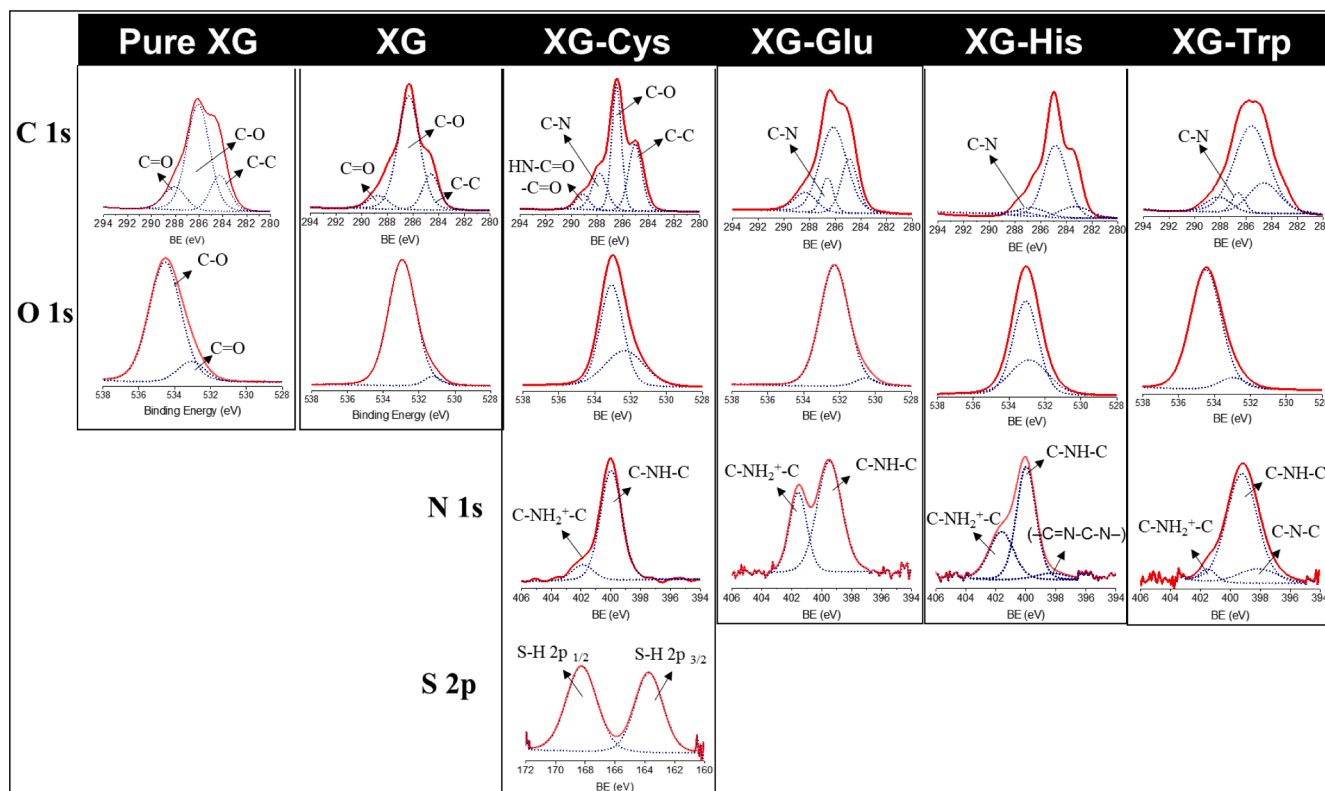
increasing the coating thickness, the influence of the intermolecular interactions became less pronounced and the chains arrangement was similar.

Table 1 shows the contact angle ( $\theta$ ) values determined for MilliQ water and  $\text{CH}_2\text{I}_2$  droplets on XG based coatings, along with the polar ( $\gamma_p\text{S}$ ) and dispersive ( $\gamma_d\text{S}$ ) components of the surface energy ( $\gamma\text{S}$ ). In comparison to pure XG, the attachment of amino acid to the XG chains did not affect the  $\gamma_d\text{S}$  component, but decreased considerably the  $\gamma_p\text{S}$  component. Thin films presented larger rms values (Table 1) than ultrathin films (Fig. 3), that probably caused deviations in the contact angle measurements of  $\sim 10\%$ . The increase of surface roughness of Ti-6Al-4 V alloys caused variation in the  $\gamma\text{S}$  values from  $45 \text{ mJ/m}^2$  to  $52 \text{ mJ/m}^2$  (Yan, Chibowski & Szcześ, 2017).

The chemical composition of thin coatings was analyzed by XPS. Fig. 5 shows the high-resolution XPS spectra for carbon (C 1 s), oxygen (O 1 s), nitrogen (N 1 s) and sulfur (S 2p) binding energy (BE) regions. The deconvolution of carbon satellite peak (C 1 s) for pure XG (no AA or



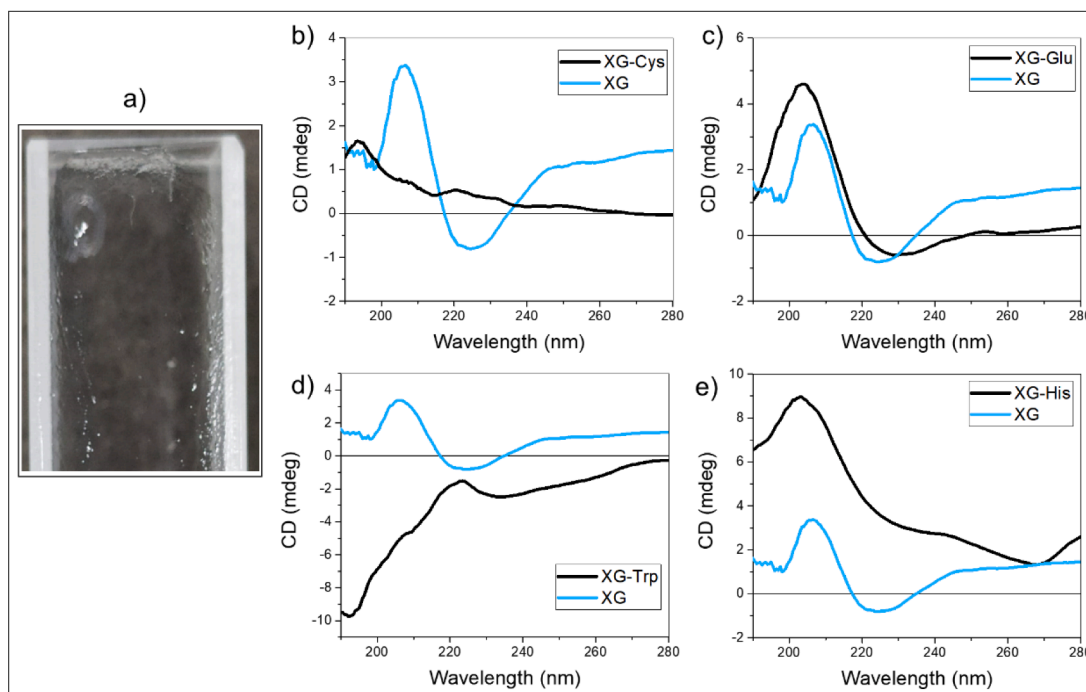
**Fig. 4.** AFM topographic images of (a) XG, (b) XG-Cys, (c) XG-Glu, (d) XG-His and (e) XG-Trp-thin films.



**Fig. 5.** Deconvolution of XPS high-resolution spectra of C 1 s, N 1 s, O 1 s and S 2 p, along with the corresponding BE values determined for pure XG, XG coatings, XG-Cys, Glu, XG-His and XG-Trp-coatings.

crosslinker) indicated three peaks attributed to C—C (284.5 eV) as dominant component, C—O (286.1 eV) and C=O (288.0 eV) bonds (Bulbul, Bhushette, Zambare, Deshmukh & Annapure, 2019). After crosslinking, XG coatings presented C 1 s peaks similar to those of pure XG; however, the intensity associated to C—O bond increased,

confirming the esterification reaction between citric acid and XG hydroxyl groups. The grafting of amino acids to XG chains was identified by the appearance of a new peak in C 1 s region, typical of C—N bonds at 287.8 eV (XG-Cys), 286.3 eV (XG-Glu), 287.7 eV (XG-His), and 286.5 eV (XG-Trp) (A. Artemenko et al., 2021). Additionally, the acetamide (O =



**Fig. 6.** (a) Photograph of a PBS swollen XG coating deposited on the cuvette for circular dichroism (CD) measurements. CD spectra of (b) XG (blue) and XG-Glu (black) coatings, (c) XG (blue) and XG-Cys (black) coatings, (d) XG (blue) and XG-His (black) coatings and (e) XG (blue) and XG-Trp (black) coatings.



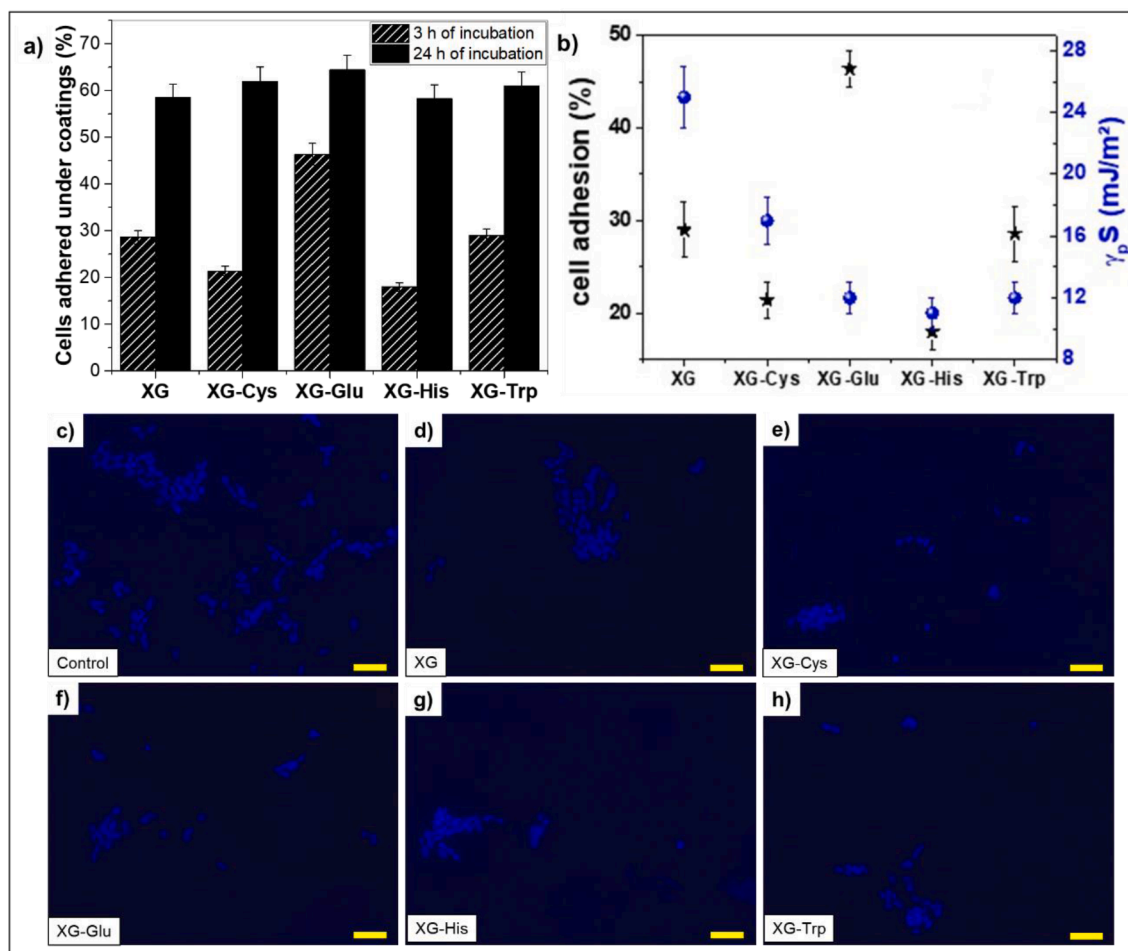
C—N) bond appears mutually in the same BE region as C—N. In the case of N 1 s region, two peaks signals were assigned to secondary protonated amine (C—NH<sub>2</sub><sup>+</sup>—C) and secondary amine (C—NH—C), due to the amide formation. A third peak for C = N at 398 eV was assigned to imidazole in XG-His-and indole in XG-Trp-coatings (A. Artemenko et al., 2021; O. Cavalleri et al., 2004; J.S. Stevens et al., 2013). XG-Cys-coatings presented doublet peaks at 164/168 eV BE related to 2p<sub>3/2</sub> and 2p<sub>1/2</sub> S-H bond, which is typical for Cys (J.S. Stevens et al., 2013). Table SM3 comprises the peaks after deconvolution of XPS high-resolution spectra of C 1 s, N 1 s, O 1 s and S 2p, along with the corresponding BE values.

XPS analyses in Fig. 5 and Supplementary Material SM9 supported the reactions mechanism evidenced by FTIR-ATR spectra (Fig. 2), where citric acid molecules mediated the crosslinking of XG chains and the grafting of amino acids. In order to investigate if these reactions affect the conformational state of XG chains, thin XG based coatings were analyzed by circular dichroism (CD). The coatings were kept swollen in PBS buffer (Fig. 6a) during the measurements. The CD spectra obtained for XG and XG-Glu-coatings (Fig. 6b) presented similar features. The positive signal at ~ 203 nm – 205 nm attributed to n → π\* transition of the carboxylate groups of XG (D-glucuronic acid and pyruvate groups) (V.B. Bueno & Petri, 2014; N.M. Eren, Santos & Campanella, 2015), and the negative signal at ~ 220 nm due to XG acetate groups (V.B. Bueno & Petri, 2014) indicated the helix conformation of XG chains in XG and XG-Glu-coatings. Thus, the attachment of Glu-to the XG chains did not affect the helix conformation of XG chains in the swollen coatings. On the other hand, grafting Cys (Fig. 6c), His (Fig. 6d) and Trp (Fig. 6e) caused substantial changes in the conformational state of XG chains in

the corresponding coatings. In the case of XG-Cys (Fig. 6c) and XG-His (Fig. 6d) coatings the positive signal in the range of 195 nm to 205 nm could be observed, but the negative signal disappeared, indicating that the helix conformation was partially lost. In the case of XG-Trp-coatings (Fig. 6e), the XG chains seem to have lost completely the ordered helix conformation, assuming random coil conformation. The changes in the molecular conformation of XG chains might be attributed to the interactions between the amino acid functional groups attached to the chains. Particularly in the case of XG-Trp-coatings, π-π interactions between Trp-indole groups can approximate XG chains segments, inducing conformational changes in the chains.

#### 3.4. Cell adhesion and viability on thin XG based coatings

The cell adhesion and viability assays were performed on thin XG-based coatings because they are more uniform than the ultrathin films and coatings thicker than 50 nm proved to be more suitable for studies on cell behavior (Bhattacharyya, Xu, Deshmukh, Timmons & Nguyen, 2010). Fig. 7a shows the adhesion of SH-5YSY cells after 3 h and 24 h of incubation in comparison to the control (100%). After 3 h incubation, the lowest cell adhesion was observed for XG-His-and XG-Cys-coatings, 18% and 21.4%, respectively. XG and XG-Trp-coatings presented the intermediate adhesion rates, namely, 29% and 28.6%, respectively, and XG-Glu-coatings showed the highest adhesion (46.4%), in comparison to the control. Fig. 7b shows that the % cell adhesion tended to decrease with the decrease of γ<sub>p</sub>S, except for XG-Glu-and XG-Trp. The adhesion of fibroblasts was very low on surfaces with low γ<sub>p</sub>S values (polyolefin,



**Fig. 7.** (a) Cell adhesion (%) on XG-based coatings in comparison to the cell adhesion on the plastic cell plate after 3 h and 24 h incubation (error bars set to 5%). (b) Cell adhesion (%) after 3 h and the polar component of surface energy (γ<sub>p</sub>S) determined for XG-based coatings. Fluorescence microscopy images (DAPI stained) of SH-5YSY cells incubated (24 h) on control (c), XG (d), XG-Cys (e), XG-Glu (f), XG-His (g) and XG-Trp (h) coatings. Scale bar: 100 μm.

Teflon, silicone rubber), but increased considerably on more hydrophilic surfaces (glass, metals stainless steel, cobalt-chromium alloy, titanium alloy and tantalum) (Hallab, Bundy, O'Connor, Moses & Jacobs, 2004). Recently, Mahjy et al. prepared poly(dimethyl siloxane) with surface energy values ranging from 21 mJ/m<sup>2</sup> to 100 mJ/m<sup>2</sup> and roughness values ranging from 5 nm to 150 nm; surfaces with  $\gamma_s$  of  $\sim 70$  mJ/m<sup>2</sup> and intermediate roughness values were the most favorable for the adhesion of cells (Majhy, Priyadarshini & Sen, 2021). Therefore, the low adhesion of cells on thin XG-based coatings can be attributed to the low roughness ( $\sim 20$  nm) and  $\gamma_s$  of  $\sim 50$  mJ/m<sup>2</sup> (Table 1). The lowest cell adhesion on XG-His- and XG-Cys-coatings after 3 h incubation might also be correlated with the partial loss of helix conformation observed in Figs. 6c and 6d.

After 24 h incubation, all XG-based coatings presented similar cell adhesion of  $\sim 60\%$  in comparison to the control (Fig. 7a). This finding indicated that the coating surface energy and chains molecular

conformational state affected the cell adhesion mainly at short incubation time (3 h). As the incubation time increases, the surface roughness might increase due to the adhesion of the first cells. Moreover, for long incubation time (24 h), the coatings became swollen by the culture medium, minimizing the effects of the surface energy components observed in the short incubation time. A similar trend was observed for the adhesion of MG-63 cells on poly(methyl methacrylate) (PMMA), polystyrene (PS), and poly(dimethyl siloxane) (PDMS) at 1 and 3 h incubation (Comelles, Estévez, Martínez & Samitier, 2010). Figs. 7c-7h show the fluorescence images of SH-SY5Y cells stained with DAPI on XG based coatings along with the control experiment after 24 h incubation. They clearly showed less pronounced cell adhesion on the XG coatings in comparison to the adhesion on the control.

In vitro biocompatibility of XG-based coatings was evaluated by MTT test with the extract solution resulting from the coating in contact with cell culture medium (24 h). After 24 h of cell incubation, none of the XG

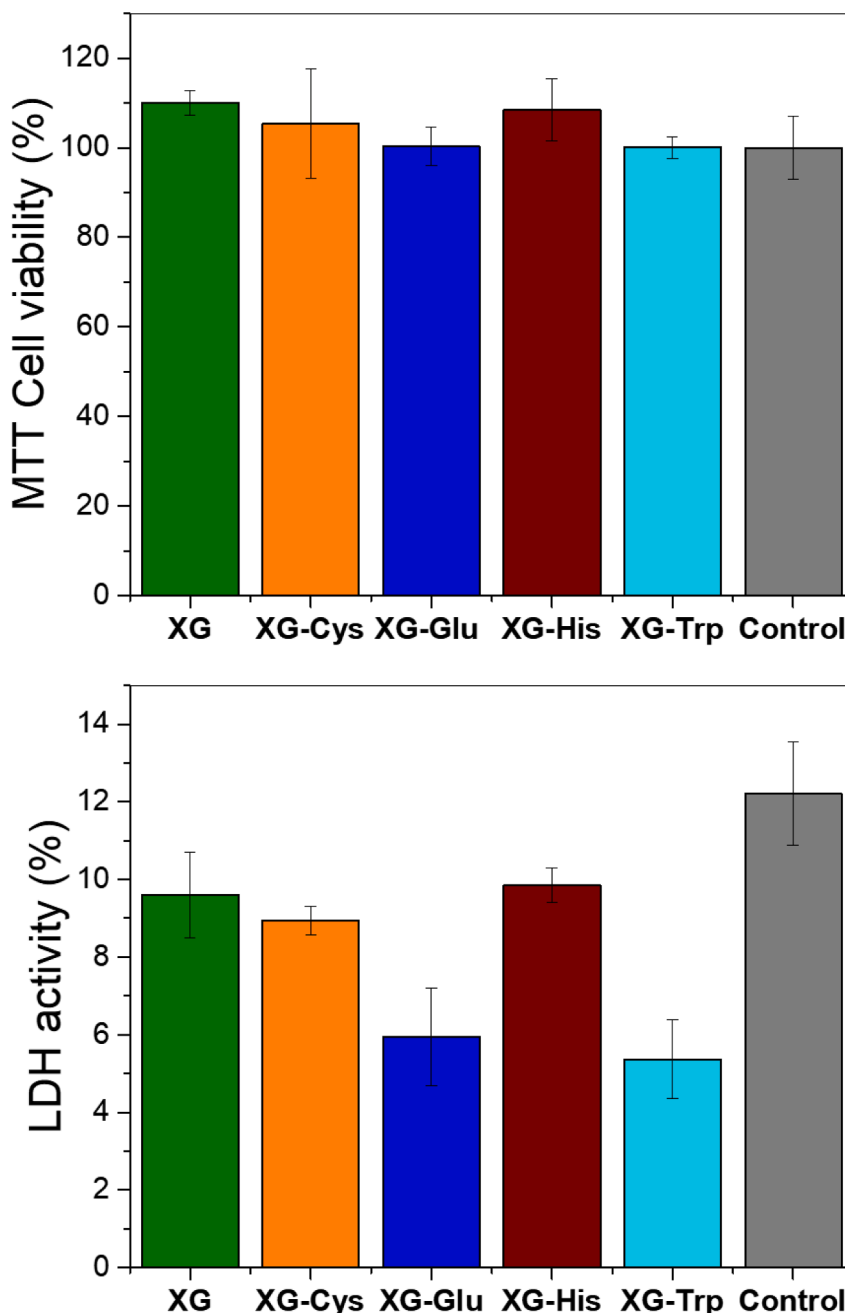


Fig. 8. (a) Cell viability (SH-SY5Y) of XG-based coatings using MTT assay and (b) Lactose dehydrogenase activity tests in the culture medium incubated for 24 h.



based coatings showed cytotoxicity (Fig. 8a). The amount of dead cells on the extract solutions was evaluated by the LDH assay (Fig. 8b). XG, XG-Cys-and XG-His-coatings showed LDH activity similar to the control; whereas the LDH values for XG-Glu-and XG-Trp-coatings were the lowest.

#### 4. Conclusions

The crosslinking of XG chains and the grafting of AA onto XG chains mediated by CA molecules were clearly evidenced by FTIR-ATR spectra and TGA. Elemental analyses indicated  $\sim 0.20$  g/g of AA grafted to XG chains. Thin ( $\sim 60$  nm) XG-amino acid coatings were successfully deposited on Si/SiO<sub>2</sub> wafers, quartz cuvettes and glass slides. In comparison to pure XG coatings, the grafting of Glu, Cys, His-or Trp-to the XG chains did not affect the dispersive surface energy ( $\gamma_d S$ ) component, but decreased considerably the polar surface energy ( $\gamma_p S$ ) component, making the adhesion of SH-SY5Y cells on them less than 30% or 60% of that observed for the control (plastic plate) after 3 h or 24 h incubation, respectively. The lowest cell adhesion after 3 h incubation observed for Cys-or His-grafted to XG coatings might also have been caused by the less ordered conformational state of chains observed in the CD spectra. The intermolecular interactions between Cys, His-and Trp-functional side groups affected the conformational state of XG chains in the corresponding coatings. Considering that none of the XG-amino acid coatings presented cytotoxicity against SH-SY5Y cells, all of them are potential candidates as coatings for neural electrodes. Nevertheless, for short incubation time, the XG-His (lowest  $\gamma_p S$  and random molecular conformational) is the most adequate to hinder cell adhesion on the coating.

#### Declaration of Competing Interest

The authors declare that they have no known competing financial interests or personal relationships that could have appeared to influence the work reported in this paper.

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#### Supplementary materials

Supplementary material associated with this article can be found, in the online version, at [doi:10.1016/j.carpta.2022.100227](https://doi.org/10.1016/j.carpta.2022.100227).

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