



Crashworthiness assessment of a composite fuselage stanchion employing a strain rate dependent damage model

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ARTICLE INFO

Keywords:
Crashworthiness
Composites
Fuselage
Damage modelling

ABSTRACT

Numerical testing is crucial for the design of composite fuselages, which have strict crashworthiness regulations. However, the majority of studies on numerical fuselage impacts do not account for the effects of strain rate in simulations. A damage model considering strain rate dependence has been implemented to accurately predict the impact behaviour of a composite fuselage structure. This model enhances the existing three-dimensional Hashin criterion by incorporating strain rate effects and its implemented numerically using a VUMAT subroutine in ABAQUS/explicit. Validation of the model is done through a low-velocity impact problem, showing a better correlation with experimental data compared to previous numerical analyses available in the literature. The study focuses on high-energy impact on a composite stanchion in the lower lobe of an aircraft fuselage. Results demonstrate that the newly proposed model effectively predicts failure zones and modes, indicating its potential in addressing dynamic composite problems typical of impact scenarios.

1. Introduction

As the aeronautical industry seeks to increase performance and optimise maintenance costs, Carbon Fibre Reinforced Plastic (CFRP) materials have been incorporated into civil aircraft such as the A350 due to their superior properties, including high specific strength, excellent resistance to corrosion, and extended fatigue life [1–3]. Airworthiness regulations require that no change in occupant safety should occur whether composite or metal structures are employed [4]. As a proof of compliance, manufacturers are required to show either by direct tests or test supported analysis (building block-approach [4,5]) that the aircraft satisfies different criteria, such as retention of items of mass (e.g. overhead bins) or maintenance of a survivable volume [4,6]. Given that experimental tests are costly, manufacturers rely on computational tools such as finite element analysis to evaluate different geometries or design variables such as different number of plies, stacking sequence or even impact orientation and therefore, it is of great interest to employ accurate models to aid the design process [4].

There are many different works on the behaviour of composite structures under impact [7–11], where it is observed that contrarily to metals where energy is absorbed by plastic deformation, the main energy composite absorption mechanisms come for intra (inside the ply) and inter-laminar (delamination) damage [12,13]. To

numerically predict these failure modes, researchers have proposed various models [14–17], having some (mostly Hashin [18] and Chang-Chang [19]) been employed in simulating both thin-walled composite columns [20–22] and composite fuselage components under compression and impact scenarios [23–29]. However, for fuselage components, sometimes a noticeable difference is observed between numerical and experimental results. One factor that is rarely present in past studies, and practically not addressed in composite fuselage related problems, is the strain rate effect. Carbon fibres are considered to be practically strain insensitive [30], however, the polymeric material is not, and can influence the behaviour of the laminate, specially under compression [31]. There have been some models proposed to account for the strain rate effect in CFRP [31–33] such as the one proposed by Millen et al. [34]. Although there are some examples in literature of works which incorporate strain rate effects in impact simulations, adjusting the material properties *a priori* [35] or by direct calculation during the analysis [36], their generalised employment in numerical simulations of impact problems is still limited, specially for fuselage components, as pointed out by Liu et al. [37] in their extensive review.

Building on this groundwork, the present study investigates the potential benefits of incorporating strain rate effects into the numerical impact analysis of composite parts and fuselage structures. A 3D Hashin

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strain rate dependent failure model based on the strategy employed by Millen et al. [34] is used in the impact simulation of composite parts. The material damage model is introduced as well as the strategy followed to include the strain rate effect. The developed model is verified using a simple one element displacement controlled test. The material model is then validated with a Low Velocity Impact (LVI) problem and bench-marked against experimental and numerical results from the literature. After validating the model, an impact of a composite stanchion is simulated, where the failure mode is studied and the results are compared to documented experimental observations.

The key novelty of this study is the incorporation of strain rate sensitivity in impact simulations of composite structures, which has been largely overlooked in previous works, specially regarding fuselage components (e.g., [23,25,27]). By introducing this feature, this work offers a more accurate and comprehensive framework for predicting material failure under dynamic conditions, which can be applied not only for aeronautical structures, but also for all problems related to crashworthiness analysis.

2. Material model

In this section, the material laws employed in this study are presented, beginning with the intra-laminar model and concluding with the inter-laminar damage model. The intra-laminar model is organised in subsections: damage onset (failure criterion), damage evolution, and the effect of strain rate.

2.1. Intra-laminar damage

2.1.1. Damage onset

To determine the damage onset, a 3D Hashin failure criterion [18] was used, being able to distinguish 4 different failure modes which are defined by the following equations:

- Fibre tensile failure ($\sigma_{11} \geq 0$)

$$F_{ft} = \left(\frac{\sigma_{11}}{X_T} \right)^2 + \alpha \left(\frac{\sigma_{12}}{S_{12}} \right)^2 + \alpha \left(\frac{\sigma_{13}}{S_{13}} \right)^2 \geq 1 \quad (1)$$

- Fibre compressive failure ($\sigma_{11} < 0$)

$$F_{fc} = \left(\frac{\sigma_{11}}{X_C} \right)^2 \geq 1 \quad (2)$$

- Matrix tensile failure ($\sigma_{22} + \sigma_{33} \geq 0$)

$$F_{mt} = \left(\frac{\sigma_{22} + \sigma_{33}}{Y_T} \right)^2 + \frac{1}{S_{23}^2} (\sigma_{23}^2 - \sigma_{22}\sigma_{33}) + \left(\frac{\sigma_{12}}{S_{12}} \right)^2 + \left(\frac{\sigma_{13}}{S_{13}} \right)^2 \geq 1 \quad (3)$$

- Matrix compressive failure ($\sigma_{22} + \sigma_{33} < 0$)

$$F_{mc} = \left(\frac{\sigma_{22} + \sigma_{33}}{2S_{23}} \right)^2 + \frac{\sigma_{22} + \sigma_{33}}{Y_C} \left[\left(\frac{Y_C}{2S_{23}} \right)^2 - 1 \right] + \frac{1}{S_{23}^2} (\sigma_{23}^2 - \sigma_{22}\sigma_{33}) + \left(\frac{\sigma_{12}}{S_{12}} \right)^2 + \left(\frac{\sigma_{13}}{S_{13}} \right)^2 \geq 1 \quad (4)$$

where σ_{ij} ($i, j = 1, 2, 3$) are the stress tensor values, X_T , X_C , Y_T and Y_C are the tensile and compressive strengths in the fibre and matrix directions respectively. The values S_{12} , S_{13} and S_{23} represent the shear strengths in different directions and α , is a parameter which controls the influence of the shear stress in tensile failure, which, in this work is set to 1 [38].

2.1.2. Degradation model

An energy based linear degradation model was employed, i.e. after damage onset, as seen in Fig. 1 the stiffness decreases linearly so that

the total energy equals the fracture energy of the material for that specific failure mode.

In Fig. 1, $\delta_{I,eq}^f$, $\delta_{I,eq}^0$ and $\delta_{I,eq}$ are the equivalent displacements at total failure, damage onset and at the respective time increment, respectively. The values of equivalent displacement and stress are dependent on the failure mode and are defined as follows [39]:

- Fibre tensile failure ($\sigma_{11} \geq 0$)

$$\delta_{eq}^{ft} = l_c \sqrt{\langle \epsilon_{11} \rangle^2 + \epsilon_{12}^2 + \epsilon_{13}^2} \quad (5)$$

$$\sigma_{eq}^{ft} = \frac{l_c (\langle \sigma_{11} \rangle \langle \epsilon_{11} \rangle + \sigma_{12} \epsilon_{12} + \sigma_{13} \epsilon_{13})}{\delta_{eq}^{ft}}$$

- Fibre compressive failure ($\sigma_{11} < 0$)

$$\delta_{eq}^{fc} = l_c \langle -\epsilon_{11} \rangle \quad (6)$$

$$\sigma_{eq}^{fc} = \frac{l_c \langle -\sigma_{11} \rangle \langle -\epsilon_{11} \rangle}{\delta_{eq}^{fc}}$$

- Matrix tensile failure ($\sigma_{22} + \sigma_{33} \geq 0$)

$$\delta_{eq}^{mt} = l_c \sqrt{\langle \epsilon_{22} \rangle^2 + \langle \epsilon_{33} \rangle^2 + \epsilon_{12}^2 + \epsilon_{23}^2 + \epsilon_{13}^2} \quad (7)$$

$$\sigma_{eq}^{mt} = \frac{l_c (\langle \sigma_{22} \rangle \langle \epsilon_{22} \rangle + \langle \sigma_{33} \rangle \langle \epsilon_{33} \rangle + \sigma_{12} \epsilon_{12} + \sigma_{23} \epsilon_{23} + \sigma_{13} \epsilon_{13})}{\delta_{eq}^{mt}}$$

- Matrix compressive failure ($\sigma_{22} + \sigma_{33} < 0$)

$$\delta_{eq}^{mc} = l_c \sqrt{\langle -\epsilon_{22} \rangle^2 + \langle -\epsilon_{33} \rangle^2 + \epsilon_{12}^2 + \epsilon_{23}^2 + \epsilon_{13}^2} \quad (8)$$

$$\sigma_{eq}^{mc} = \frac{l_c \langle -\sigma_{22} \rangle \langle -\epsilon_{22} \rangle + \langle -\sigma_{33} \rangle \langle -\epsilon_{33} \rangle + \sigma_{12} \epsilon_{12} + \sigma_{23} \epsilon_{23} + \sigma_{13} \epsilon_{13})}{\delta_{eq}^{mc}}$$

where l_c is the characteristic length, defined by the cubic root of the element volume and $\langle \cdot \rangle$, the Macaulay operator, defined as:

$$\langle a \rangle = \frac{a + |a|}{2}, \forall a \in \mathbb{R} \quad (9)$$

To follow the bi-linear behaviour, each element stiffness matrix needs to be changed once damage is initiated and is represented by the following expression:

$$[C_d] = \frac{1}{A} \begin{bmatrix} C_{11} & C_{12} & C_{13} & 0 & 0 & 0 \\ C_{21} & C_{22} & C_{23} & 0 & 0 & 0 \\ C_{31} & C_{32} & C_{33} & 0 & 0 & 0 \\ 0 & 0 & 0 & C_{44} & 0 & 0 \\ 0 & 0 & 0 & 0 & C_{55} & 0 \\ 0 & 0 & 0 & 0 & 0 & C_{66} \end{bmatrix} \quad (10)$$

where the subscript d indicates that it is a damaged stiffness matrix, which following the approach used by Zhou et al. [39] (based on Matzenmiller et al. [40]) is defined as follows:

$$C_{11} = d_f E_{11} (1 - d_m v_{23} v_{32}) \quad (11)$$

$$C_{22} = d_m E_{22} (1 - d_f v_{13} v_{31})$$

$$C_{33} = E_{33} (1 - d_f d_m v_{12} v_{21})$$

$$C_{12} = d_f d_m E_{11} (v_{21} + v_{23} v_{31})$$

$$C_{13} = d_f E_{11} (v_{31} + d_m v_{21} v_{32})$$

$$C_{23} = d_m E_{22} (v_{32} + d_f v_{12} v_{31})$$

$$C_{44} = \Lambda d_f d_m G_{12}$$

$$C_{55} = \Lambda d_f d_m G_{23}$$

$$C_{66} = \Lambda d_f d_m G_{13}$$

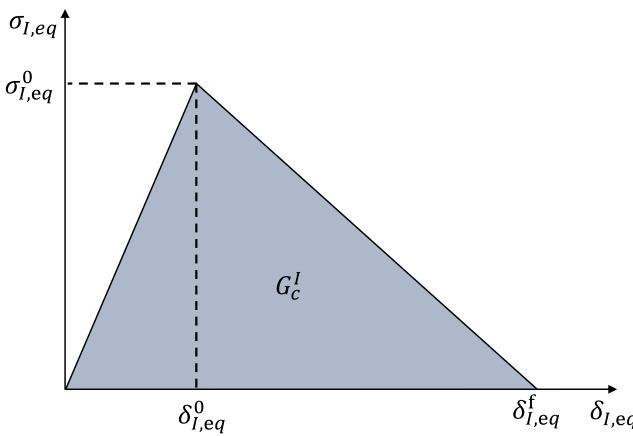


Fig. 1. Linear degradation model.

being the damage variables defined as

$$\begin{cases} d_f = (1 - d_{ft})(1 - d_{fc}) \\ d_m = (1 - S_{mt}d_{mt})(1 - S_{mc}d_{mc}) \\ \Lambda = 1 - d_f d_m v_{12} v_{21} - d_m v_{23} v_{32} - d_f v_{13} v_{31} - 2d_f d_m v_{21} v_{32} v_{13} \end{cases} \quad (12)$$

where $d_{ft}, d_{fc}, d_{mt}, d_{mc}$ represent different damage variable associated with different failure modes i.e. fibre (aligned with the fibres) and matrix (normal to the fibres) tensile and compressive failure respectively, which values can range from 0 (undamaged) to 1 (fully damaged) and their evolution is given by the following equation:

$$d_I = \frac{\delta_{I,eq}^f(\delta_{I,eq} - \delta_{I,eq}^0)}{\delta_{I,eq}(\delta_{I,eq}^f - \delta_{I,eq}^0)} \quad (d_I \in [0, 1], I = ft, fc, mt, mc) \quad (13)$$

The values of S_{mt} and S_{mc} are shear stiffness loss control parameters which are set to 0.9 and 0.5 respectively, based on [38].

Given that total failure occurs whenever the total fracture energy is achieved, the equivalent displacement at total failure is determined by:

$$\delta_{I,eq}^f = \frac{2G_C^I}{\sigma_{I,eq}^0} \quad (I = \{ft, fc, mt, mc\}) \quad (14)$$

where G_{Ic} is the energy release rate of each failure mode, I , and $\sigma_{I,eq}^0$ is the equivalent stress level at damage onset, which, in this work, will be defined by the expressions below [39]:

$$\delta_{I,eq}^0 = \frac{\delta_{I,eq}}{\sqrt{F_I}} \quad (15)$$

$$\sigma_{I,eq}^0 = \frac{\sigma_{I,eq}}{\sqrt{F_I}} \quad (16)$$

where F_I represents the failure index value of the respective failure mode and thus takes values near unity at damage onset.

2.1.3. Strain rate effect

To account for the strain rate effect, the approach used by Millen et al. [34] is considered. It consists on using strain rate dependent scaling factors which act on some mechanical properties. Following Millen et al. [34], three different regimes are considered, quasi-static, 0.1 s^{-1} , high strain rate, 100 s^{-1} , and the intermediate region. From the experimental data available in the literature, the authors were able to determine some mechanical properties for the lower and upper values of strain rate, linearly interpolating the property in the intermediate region following the expression:

$$\beta = \begin{cases} 1.0 & \text{if } \dot{\epsilon}_2 \leq 0.1 \\ \frac{(\dot{\epsilon}_2 - 0.1)(1.29 - 1)}{100 - 0.1} + 1 & \text{if } 0.1 < \dot{\epsilon}_2 < 100 \\ 1.29 & \text{if } \dot{\epsilon}_2 \geq 100 \end{cases} \quad (17)$$

Table 1
Maximum scaling factors ($\dot{\epsilon}_2 \geq 100$) [34].

	E_2	G_C^{mc}	G_C^{mt}	X_C	X_T	Y_C	Y_T
Scaling Factor	1.29	1.63	1.23	1.39	1.38	1.37	1.38

where β is the scaling parameter which adjusts the property such as, for instance the transversal stiffness:

$$E_{2f} = \beta E_2 \quad (18)$$

being E_{2f} the effective transversal stiffness. The scaling factors used in this work, based on the work of Millen et al. [34], are shown in Table 1.

For each time step, t , the strain rate is calculated by dividing the strain increment by the time increment size, Δt :

$$\dot{\epsilon}^t = \frac{\epsilon^t - \epsilon^{t-\Delta t}}{\Delta t} \quad (19)$$

after which, the scaling parameter to be applied is calculated using Eq. (17), replacing the already in-place maximum scaling factor (1.29) by the one relative to the property being considered, found in Table 1.

2.2. Inter-laminar damage

Inter-laminar damage or delamination is an important failure mode which can significantly degrade the mechanical behaviour of composite parts, mainly when impact is involved [41,42]. In this work, cohesive elements are used to simulate the inter-laminar damage and similarly to the intra-laminar damage behaviour, it also presents an onset and evolution of damage. A quadratic failure based criterion was employed to determine the damage onset:

$$\left\{ \frac{\langle t_n \rangle}{t_n^0} \right\}^2 + \left\{ \frac{t_s}{t_s^0} \right\}^2 + \left\{ \frac{t_t}{t_t^0} \right\}^2 = 1 \quad (20)$$

being t_i ($i = n, s, t$) the traction forces in the normal and both shear directions to the delamination area respectively while the denominators are each direction corresponding force allowable. Regarding the softening phase i.e. the damage evolution, the damage parameter is determined by the following equation [43]:

$$D = \frac{\delta_m^f(\delta_m^{max} - \delta_m^0)}{\delta_m^{max}(\delta_m^f - \delta_m^0)} \quad (21)$$

where δ_m^0 is the effective displacement at damage initiation, i.e. the value of δ_m^{max} when Eq. (20) is satisfied, and δ_m^f is the effective displacement at separation. The maximum equivalent or mixed-mode displacement, δ_m^{max} , is defined as

$$\delta_m^{max} = \max \left\{ \sqrt{(\delta_n)^2 + \delta_s^2 + \delta_t^2} \right\} = \max \left\{ \sqrt{(\delta_n)^2 + \delta_{shear}^2} \right\} \quad (22)$$

and being an energy based degradation failure, the effective displacement at total failure is determined by

$$\delta_m^f = \frac{2G_C}{T_m^0} \quad (23)$$

where G_C is the mixed-mode energy release rate and T_m^0 the effective traction, given by:

$$T_m^0 = K \delta_m^0 \quad (24)$$

being K the penalty stiffness. Fig. 2 shows the different mentioned values.

The mixed mode energy release rate, G_C , can be estimated using different strategies. One is the power law criterion [44]:

$$\left\{ \frac{G^n}{G_C^n} \right\}^2 + \left\{ \frac{G^s}{G_C^s} \right\}^2 + \left\{ \frac{G^t}{G_C^t} \right\}^2 = 1 \quad (25)$$

being $G_C = G^n + G^s + G^t$ when the equation is satisfied, or the Benzeggagh and Kenane (B-K) [45] model:

$$G_C = G_C^n + (G_C^s - G_C^n) \left\{ \frac{G^s}{G^T} \right\}^\eta \quad (26)$$

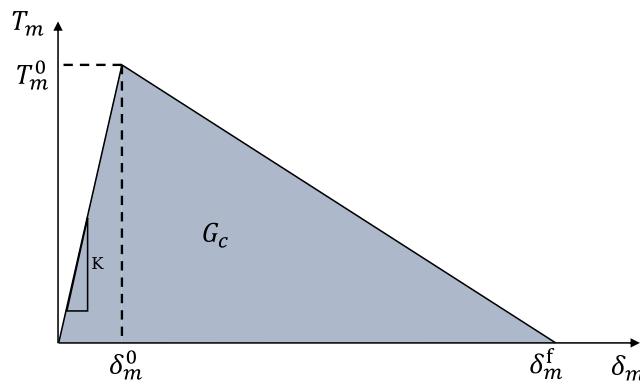


Fig. 2. Inter-laminar separation model.

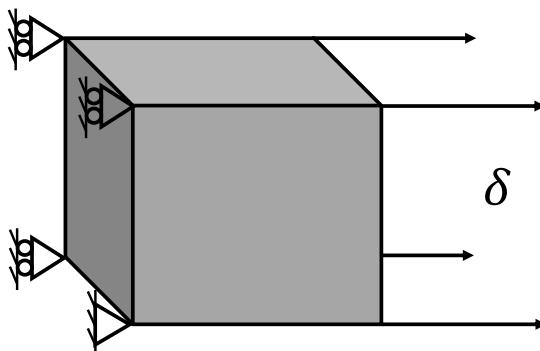


Fig. 3. One element test boundary conditions.

where $G^S = G^s + G^t$, $G^T = G^n + G^S$ and η is an experimentally obtained parameter which accounts for the mode ratio.

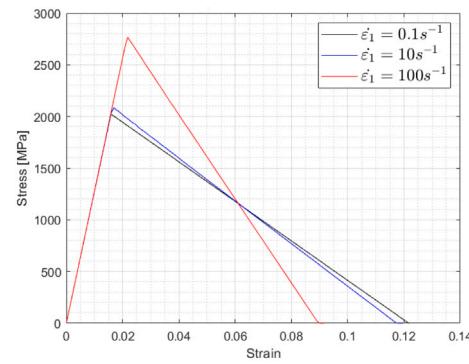
3. Numerical implementation and verification

For impact and crash problems, explicit methods are more cost efficient and therefore, explicit FE analyses were developed using Abaqus commercial software. The damage model was implemented using an user-defined material also known as VUMAT subroutine.

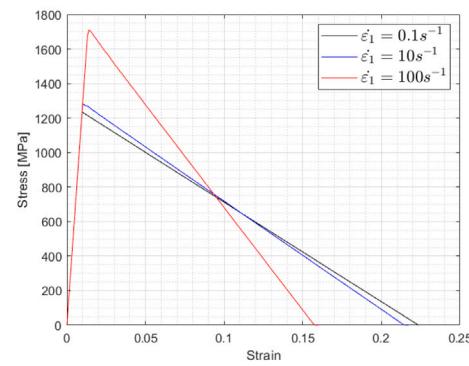
Once the code was created, a one element test was preformed for each failure mode individually. The test was carried out for a 1 mm sided cubic C3D8R (three dimensional, 8-noded linear brick element with reduced integration) where for each direction, longitudinal and transverse to the fibres. One of the normal faces is constrained on the load direction and a finite displacement, δ , is imposed at the opposite face, as seen in Fig. 3.

To ensure that failure was achieved for each failure mode, a 10 mm displacement was applied in each direction. Additionally, a node of the constrained face was fixed in order to prevent translational free body movement. Given that the tested failure criterion is strain rate dependent, three different strain rates were applied: quasi-static, intermediate and high strain rate were tested at 0.1 s^{-1} , 100 s^{-1} and 10 s^{-1} respectively, being the difference between them the simulation time to achieve the 10 mm displacement. The obtained bi-linear curves are shown in Fig. 4.

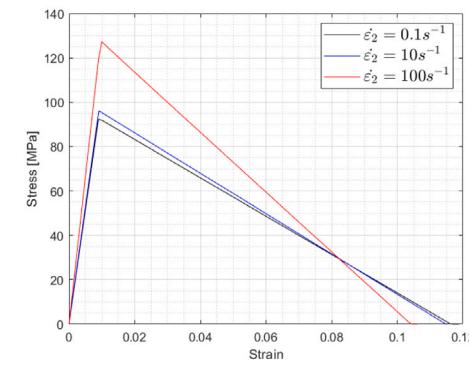
As seen from Fig. 4, there is no change in stiffness before damage onset in the longitudinal direction with the change in strain rate whereas the same does not apply to the transverse direction. Furthermore, it is observed that the strain to failure in the longitudinal direction decreases with the increase in strain rate, keeping the area enclosed by the two lines constant for all the different regimes. However, in the transverse directions the energy release rate at fracture changes with the strain rate, and therefore, the failure to strain may increase.



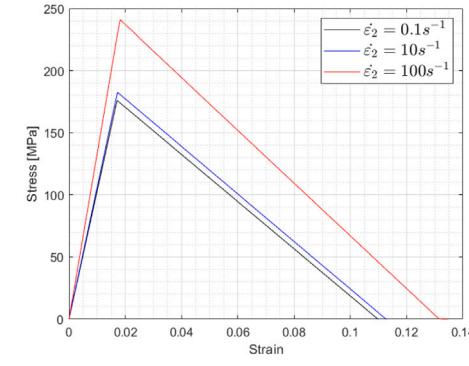
(a) Longitudinal tension



(b) Longitudinal compression



(c) Transverse tension



(d) Transverse compression

Fig. 4. Bi-linear stress-strain curves at different strain rates.

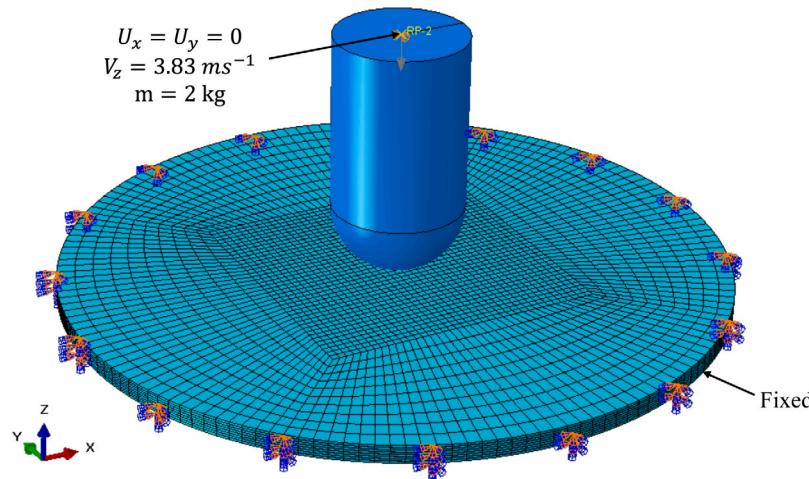


Fig. 5. Boundary conditions of the low velocity impact model.

4. Material model validation: Low velocity impact

The material model is validated against a well known low velocity impact problem previously studied by different authors [38,39,46].

4.1. Low velocity impact model

The model is a cylindrical plate with a diameter of 75 mm and a cross ply [0/90]_{2S} layup, with a layer thickness of 0.25 mm. Additionally, seven cohesive interfaces 0.0075 mm thick were added between each ply. The impactor has a 15 mm diameter hemispherical head and is modelled as an analytical rigid body to reduce simulation costs. The impactor is constrained in all directions but the vertical (Z) to ensure perpendicularity, has a mass of 2 kg and an initial velocity of 3.83 ms⁻¹, for a total impact energy of 14.7 J. The composite plate is clamped at its outer edge as seen in Fig. 5.

The geometry was partitioned to obtain a fully structured mesh with a finer area of 1 × 1 mm brick elements around the impact zone. This mesh size was chosen after a preliminary mesh sensitivity study to obtain a good compromise between mechanical response prediction accuracy and computational time. The mesh size was also aligned with previous studies in the literature [38,46], facilitating a more effective comparison between material models. In total there are 25120 C3D8R elements belonging to the plies and 21980 COH3D8 representing the cohesive interfaces. Table 2 and Table 3 list the material properties of both plies and cohesive elements respectively.

Given that reduced integration elements are being used, a 'RELAX STIFFNESS' hourglass algorithm [43] with a scaling factor, s , set to 1.5 was used, having been selected through an evaluation process considering several metrics ranging from obtained artificial strain energy and computational time. Distortion control and the limitation of the maximum degradation of the cohesive elements to 0.99 were employed to prevent extreme deformation. A general contact algorithm is employed for all parts and plies, including self-contacts, featuring 'hard' normal behaviour and penalty-type tangential behaviour. The friction coefficient has been set to 0.5 [38].

4.2. Discussion of results

The obtained results (energy-time, force-time, inter and intra-laminar damage) are compared to numerical [38,39,46] and experimental [46] data available in the literature in order to evaluate the proposed damage model.

Table 2
Material data for numerical impact validation [46].

Data	Value
Density	1600 kg/m ³
E_1 (Longitudinal Young's Modulus)	153.0 GPa
$E_2 = E_3$ (Transverse Young's Modulus)	10.3 GPa
$\nu_{12} = \nu_{13}$ (Poisson's ratio)	0.3
ν_{23} (Poisson's ratio)	0.4
$G_{12} = G_{13}$ (Shear Modulus)	6.0 GPa
G_{23} (Out-of-plane Shear Modulus)	3.7 GPa
X_T (Longitudinal Tensile Strength)	2537.0 MPa
X_C (Longitudinal Compressive Strength)	1580.0 MPa
Y_T (Transverse Tensile Strength)	82.0 MPa
Y_C (Transverse Compressive Strength)	236.0 MPa
$S_{12} = S_{13}$ (Shear Strength)	90.0 MPa
S_{23} (Out-of-plane Shear Strength)	40.0 MPa
$G_C^{f_i}$ (Fibre tensile fracture energy)	91.6 kJ/m ³
$G_C^{c_i}$ (Fibre compressive fracture energy)	79.9 kJ/m ³
$G_C^{m_i}$ (Matrix tensile fracture energy)	0.22 kJ/m ³
$G_C^{f_i}$ (Matrix compressive fracture energy)	1.1 kJ/m ³

Table 3
Material data of the cohesive elements for numerical impact validation [46].

Data	Value
Density	1600 kg/m ³
$E_n = E_s = E_t$ (Young's Modulus)	5000.0 MPa
$T_n = T_s = T_t$ (Normal and shear strengths)	30.0 MPa
G_C^n (Mode I Fracture toughness)	0.6 N/mm
$G_C^s = G_C^t$ (Mode II and III Fracture Toughnesses)	2.1 N/mm
η (Mode coupling ratio)	1.45

4.2.1. Energy and force evolution

To obtain the energy absorbed by the plate, the kinetic energy of the impactor is subtracted from the total energy:

$$E_{impactor} = \frac{m}{2} v(t)^2 \quad (27)$$

$$E_{plate} = E_{total} - E_{impactor} \quad (28)$$

where $v(t)$ is the impactor velocity. Fig. 6, shows the comparison between the numerical simulation results and the experimental and numerical analysis performed by Shi et al. [46] and Shao et al. [38] respectively.

As observed in Fig. 6, the proposed model was not only able to replicate the characteristic energy evolution of this impact problem but also to obtain a better prediction relatively to the experimental results. Analysing the initial part of the curve ($t \leq 1$ ms), representing the contact of the impactor with the plate, it is possible to observe that the obtained

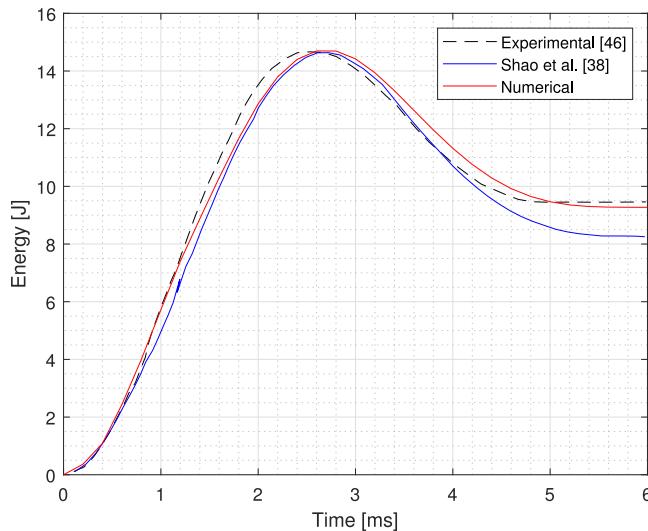


Fig. 6. Energy-time curves.

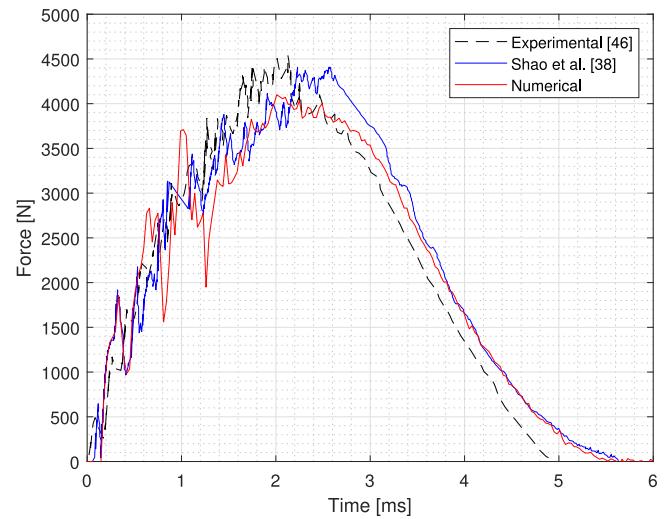


Fig. 7. Force-time curves.

Table 4
Literature and obtained absorbed energy values.

Studies	Experimental [J]	Numerical [J]	Error (%)
Shi et al. [46]		9.08	4.62
Shao et al. [38]		8.25	13.34
Zhou et al. [39]	9.52[46]	8.85	7.04
Present study		9.27	2.63

energy is closer to the experimental value, this can be explained by the increased stiffness that the strain rate dependent model brings, leading to an increased value of internal energy compared to other models. After all the kinetic energy is absorbed by the sample (peak), part of the elastic energy absorbed by the plate is returned to the impactor, decreasing the stored energy and reaching a plateau near the 5 ms mark. This value, represents the total impact energy absorbed the composite plate, which the numerical model was able to closely predict. Table 4 lists different absorbed energies obtained by different studies. It can be observed that the present study managed to achieve the better prediction relatively to the experimental value.

One other common metric in this type of problem is the force-time response which is obtained by summing the reaction force in the impact direction (z-axis) of all the fixed nodes of the plate. This curve is characterised by an initial steep increase in reaction force due to the impact which then peaks and goes back to zero, known as the rebound phase of the impactor as shown in Fig. 7.

Although there is an around 9% under-prediction of the maximum force value, it can be observed that there is a good agreement between the obtained numerical and the experimental results, especially on the peak time and rebound phase.

4.2.2. Damage area analysis

The damage distribution of different failure modes is presented and compared to previous models and experimental results. Fig. 8 compares the numerically predicted delaminated area to that obtained experimentally by Shi et al. [46] showing a good agreement between the two. The delamination damage, i.e. D of Eq. (21), is presented as the SDEG (Stiffness Degradation) parameter.

The distribution of matrix tensile damage was also compared to previously obtained numerical results to benchmark the proposed damage model (Fig. 9).

The overall shape of the damaged zones aligns with findings from previous studies, yet notably, some layers exhibit slightly smaller damaged areas, which is counterintuitive given the higher energy absorption levels observed in the energy-time evolution. This discrepancy can be attributed to the strain rate effect. At higher strain rates, the matrix stiffness and fracture energy increase, leading to a smaller damaged area for the same level of absorbed energy.

Building on previous studies and literature, these findings validate the effectiveness of the developed damage model in accurately predicting impact-related engineering problems.

5. Material model application: Stanchion impact simulation

The principal subject of analysis in this work is a composite stanchion (part of the fuselage floor support), which has previously been studied numerically and experimentally by several researchers [47–49].

5.1. Finite element model

The stanchion has a C shaped cross section with geometric details depicted in Fig. 10.

The total thickness is 2.79 mm being the layup [45/-45/90/-45/45/0/0/0/0/45/-45/90/-45/45] with 0.005 mm cohesive element interfaces between each ply. In order to have a better direct comparison with previous numerical results, the power law criterion (Eq. (25)) was used, instead of the B-K. The lamina and cohesive element properties are shown in Tables 5 and 6 respectively.

A structured mesh with 4 mm three-dimensional linear brick elements with reduced integration (C3D8R), capable of capturing a three-dimensional stress state, were employed for each composite ply, accompanied by COH3D8 for the cohesive interfaces. Given that the experimental specimen [48] was potted in epoxy resin at its extremities (at a depth of 25.4 mm each), the outer nodes of those zones were enforced a $U_x = U_y = 0$ translational constraint, while the bottom face was fixed (Fig. 11).

The contact algorithm was the same as the one previously employed in the circular plate impact and all the intra-laminar damage variables were limited to 0.95 to prevent numerical issues. A quasi-static compression test was performed to assess if the stiffness was in accordance with experimental results. Table 7 lists the stiffness calculated with the quasi static compression analysis.

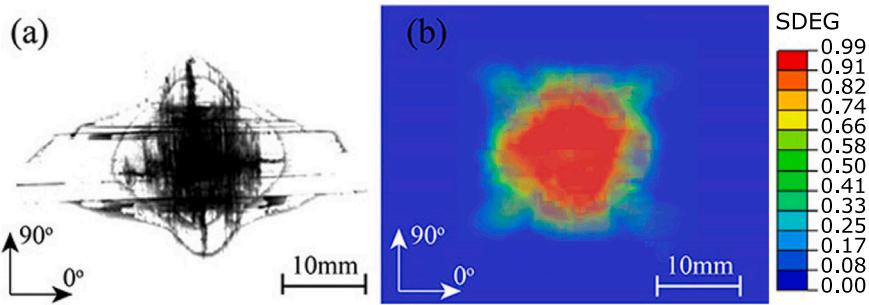


Fig. 8. Delamination distribution: (a) X-ray radiograph and (b) numerically predicted (SDEG as delamination damage variable).
Source: Adapted from [38].

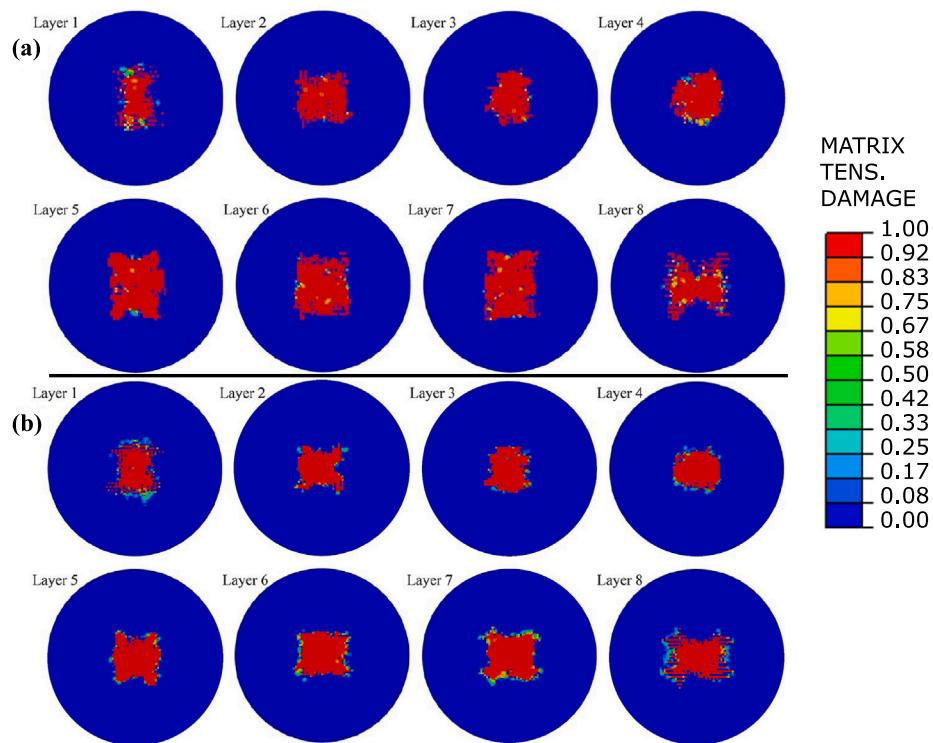


Fig. 9. Matrix tensile damage for each composite layer: (a) Shao et al. (b) numerically predicted.
Source: Adapted from [38].

5.2. Impact test

To simulate a high energy impact, a 15 ms crash event was simulated, where a plate shaped impactor, modelled as an analytical body, was set 0.3 mm away from the top face. The impactor had a mass of 77 kg and an initial vertical velocity of 3.9 ms^{-1} resulting in an impact energy of 585 J (Fig. 12).

Given that manufacturing processes are not perfect and can significantly change the behaviour of shells under compression [50], a linear buckling analysis was run and the first buckling mode multiplied by a factor of 0.05 was introduced as initial perturbation to the structure.

5.3. Results and discussion

The deformed shapes of the structure at different time steps are shown in Fig. 13.

From a preliminary observation it can be stated that the failure of the stanchion occurred near the bottom potting region where significant level of delamination occurred. From a more detailed analysis, it was observed that 0.6 ms after impact, the fibres of the 0° central block of the laminate completely failed under compression due to the high crushing loads, as shown in Fig. 14, where the fibre compression damage variable achieves its maximum mainly in the middle of the laminate. Moreover, complete separation took place between the 0° and the 45° plies due to the high shear stresses induced by the significant difference in stiffness.

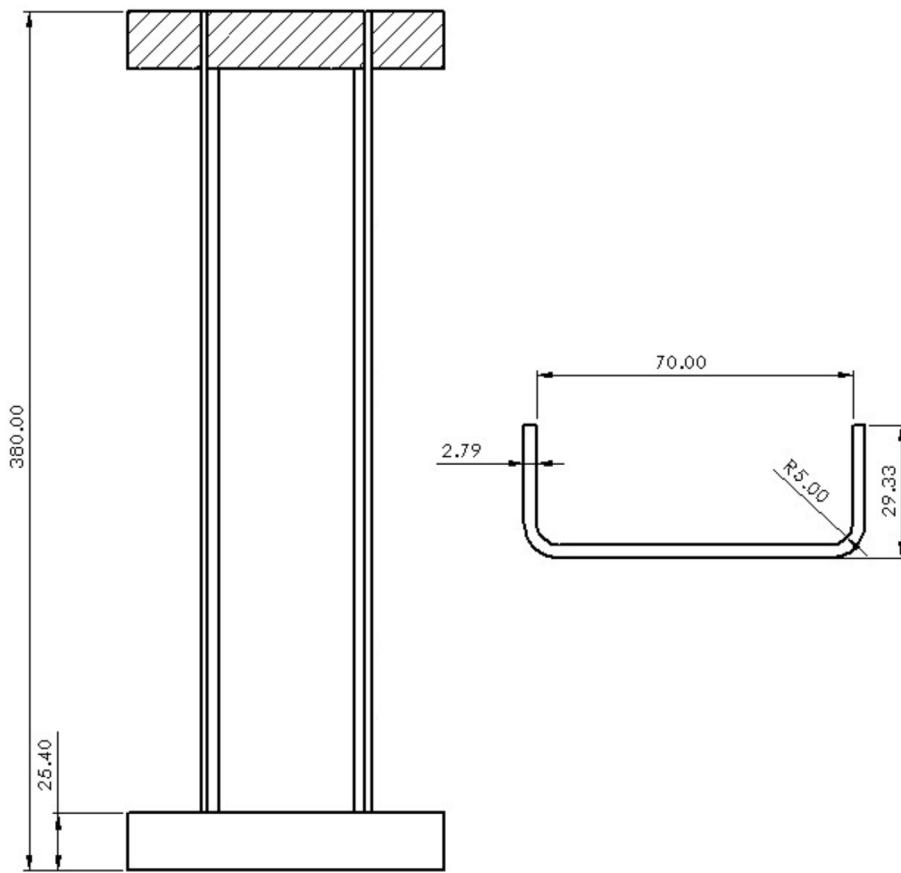


Fig. 10. Stanchion geometry (dimensions in [mm]).

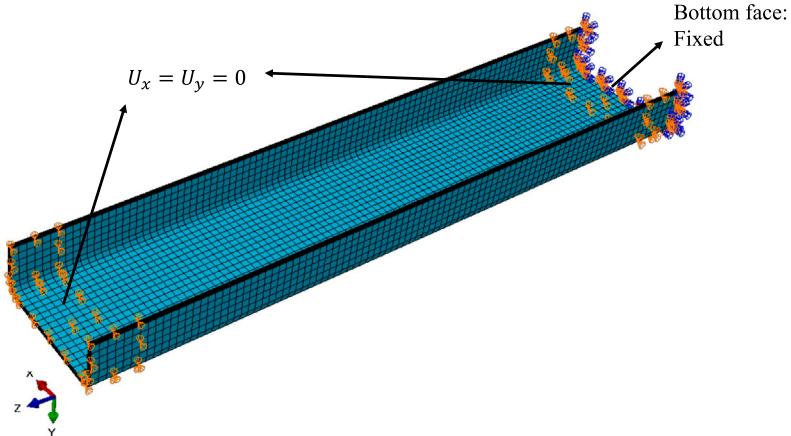


Fig. 11. Boundary conditions of the stanchion model.

The continuing crushing force then leads to the matrix compression failure of the outer sets of plies, which are the remaining load carrying plies after the initial failure of the central block. A graph showing the different damage variables through the thickness at $t=0.825$ ms after impact is shown in Fig. 15. The generalised loss of stiffness promotes a local buckling behaviour (documented failure mode of composite c-stanchions [51]), which is already noticeable to some degree in Fig. 15(a), after which, the shear induced delaminations propagate due to ply buckling and the structure practically entirely loses load carrying capabilities.

5.3.1. Comparison with experimental results

To evaluate the predictive capabilities of the developed model, a comparison with real test data is performed. The experimental results after impact are shown in Fig. 16(a), where a significant amount of damage can be seen near the potting region. Alternatively, the obtained numerical results are shown in Fig. 16(b), where the elements inside the bottom potting region were hidden for the sake of clarity.

As seen in Fig. 16, there is an excellent agreement in relation to the critical zone, where in the numerical mode is characterised by the aforementioned local buckling onset already evident in Fig. 16(b). Furthermore, recalling the damage distribution in Fig. 15, it is possible

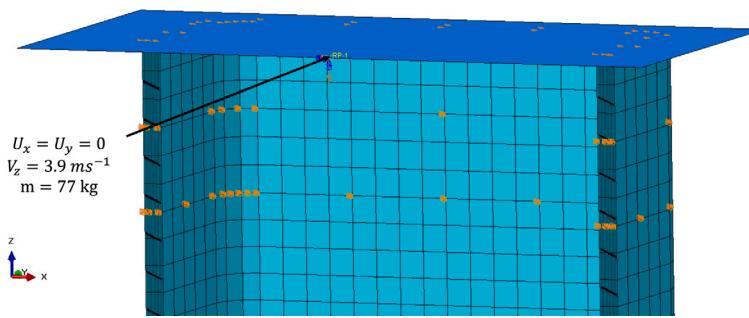


Fig. 12. Rigid impactor.

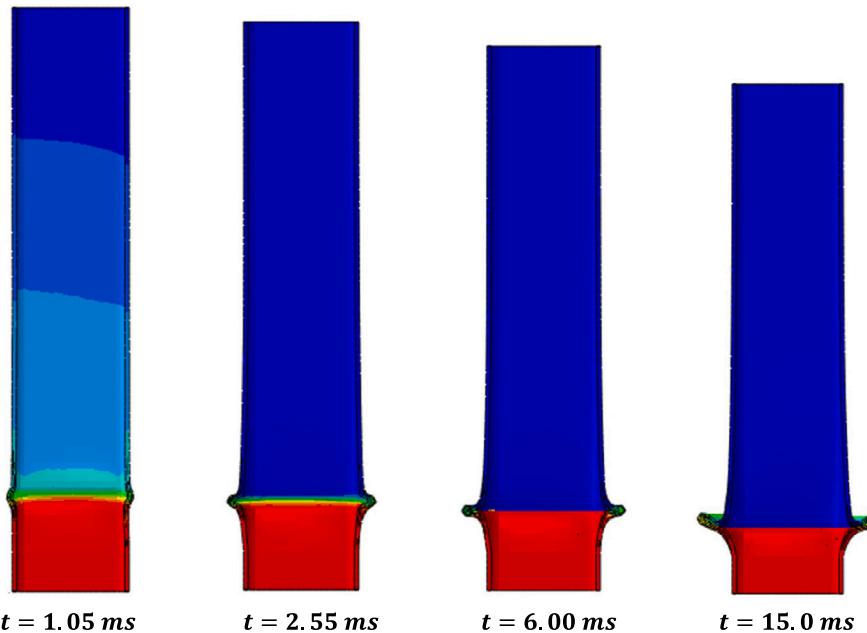


Fig. 13. Deformation at different time steps.

Table 5
Material data of the stanchion laminae [47].

Data	Value
Density	1600 kg/m ³
E_1 (Longitudinal Young's Modulus)	137.0 GPa
$E_2=E_3$ (Transverse Young's Modulus)	8.43 GPa
$\nu_{12} = \nu_{13}$ (Poisson's ratio)	0.26
ν_{23} (Poisson's ratio)	0.4
$G_{12}=G_{13}$ (Shear Modulus)	4160.0 MPa
G_{23} (Out-of-plane Shear Modulus)	3328.0 MPa
X_T (Longitudinal Tensile Strength)	2256.0 MPa
X_C (Longitudinal Compressive Strength)	800.0 MPa
Y_T (Transverse Tensile Strength)	100.0 MPa
Y_C (Transverse Compressive Strength)	171.0 MPa
$S_{12}=S_{13}$ (Shear Strength)	85.0 MPa
S_{23} (Out-of-plane Shear Strength)	40.0 MPa
G_C^{f1} (Fibre tensile fracture energy)	47.58 kJ/m ³
G_C^{f2} (Fibre compressive fracture energy)	6.01 kJ/m ³
G_C^{t1} (Matrix tensile fracture energy)	1.53 kJ/m ³
G_C^{t2} (Matrix compressive fracture energy)	4.48 kJ/m ³

Table 6
Material data of the cohesive interfaces of the stanchion [47].

Property	Value
Density	1600 kg/m ³
$E_n=E_s=E_t$ (Young's Modulus)	1000000 MPa
T_n (Normal Strength)	27.5 MPa
$T_s = T_t$ (Shear Strengths)	60.0 MPa
G_C^n (Mode I Fracture Toughness)	0.288 N/mm
$G_C^{f1} = G_C^{f2}$ (Mode II and III Fracture Toughnesses)	0.61 N/mm

Table 7
Stanchion stiffness [49].

Studies	Experimental [kN/mm]	Numerical [kN/mm]	Error (%)
Sellitto et al. [49]	54.8 [49]	56.3	2.6
Present study		54.673	0.23

to observe that some damage modes were accurately predicted, such as delaminations and fibre breakage.

Comparing the obtained numerical results (Fig. 17) with other results from the literature (Fig. 18), it is observed that instead of a

diffused damaged area, a highly localised failure zone was predicted, closer resembling experimental results. The different deformed shape is due to the local buckling phenomenon, which was not registered in the previous study.

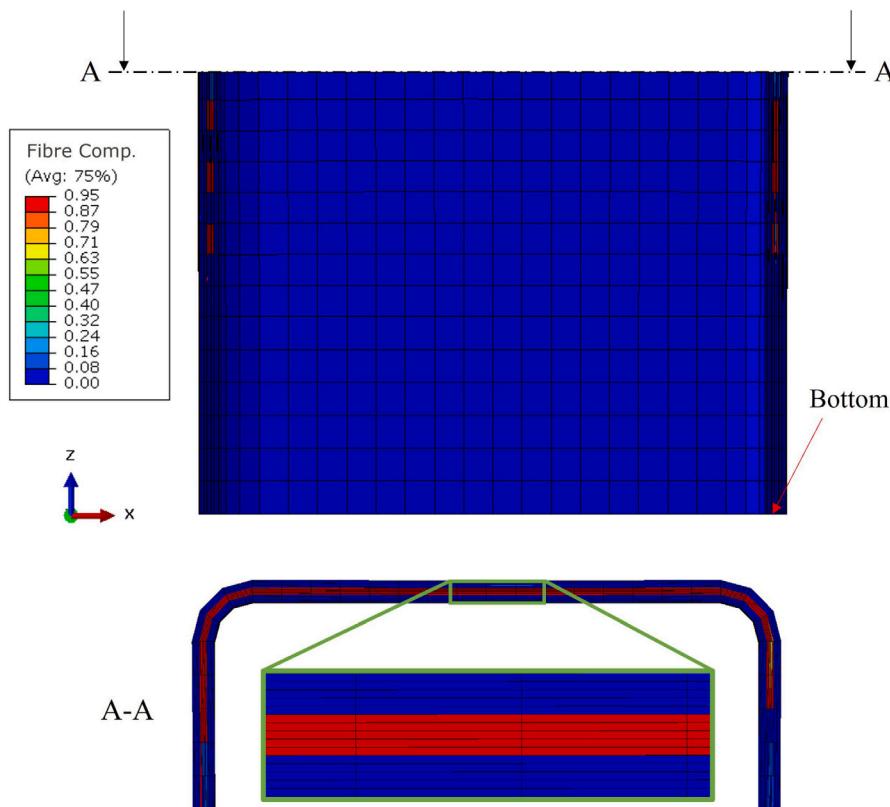


Fig. 14. Initial fibre compression failure at the central ply block ($t = 0.6$ ms after impact).

Although the results obtained differ significantly from those in the literature [47], it is not certain that this discrepancy can be entirely attributed to the inclusion of the strain rate. This uncertainty arises because different elements, such as continuum shell and different number of element through the thickness, were used in the previous analysis. The used material model is also not exactly the same, which can add to those differences.

5.3.2. Strain rate effect

To evaluate the influence of the strain rate, the same simulation was run neglecting its effect i.e. setting the values of the scaling parameters to 1. Figs. 19 and 20 compare the fibre and matrix damage distribution at the bottom of the stanchion, respectively, at a specific time step, where it is possible to observe that there is a significant damage severity increase, when considering the effect of the strain rate. This change in damage distribution can be explained by a stiffening effect [52] caused by the strain rate inclusion, leading to the embrittlement of the structure [53].

It can also be observed that the local buckling onset zone changes when using a strain insensitive failure model, which is due to the change of the spot with higher generalised loss of stiffness. Therefore, leading to a less accurate prediction relatively to the strain rate dependent model, which presented excellent correlation to the experimental results.

5.3.3. Crashworthiness metrics

In order to assess the crashworthiness performance of the stanchion and the effect of the strain rate in the simulation, the impactor kinetic energy and acceleration are plotted in Figs. 21(a) and 21(b) respectively, being the latter compared to the experimental results obtained by Riccio et al. [47].

The initial loss of kinetic energy is similar in both models and comes from its conversion into strain and fracture energy of the stanchion. After losing load carrying capabilities (around 1 ms after impact), the

kinetic energy loss decreases at a lower rate, due to the buckling of the structure. However, the rate at which the kinetic energy decreases is not the same between models. This difference is due to the different local buckling onset zones and the different deformation behaviour they cause, leading to a more progressive and stable evolution of the non-strain rate dependent model, which is then able to absorb more energy than a structure with unstable failure. The maximum acceleration value was obtained when employing the strain rate dependent model, 148 g, which is approximately 24% less than the experimental predicted value, however, it is closer than the one obtained employing the non-strain rate dependent model, where a value of 141 g was registered. This difference is attributed to the already mentioned stiffening effect. Additionally, it was found that these values were significantly sensible to the number of field output frequency, meaning that numerical time sampling under-prediction might have taken place.

One other common metric in crashworthiness is the force-displacement [51], which is plotted in Fig. 22.

The force-displacement curves reveals a sharp peak followed by a quick decline, ultimately stabilising around 12 kN. Notably, the strain rate dependent model reaches a maximum value of 93.75 kN, which is predictably greater than that of the strain rate insensitive model, 90.39 kN, due to the highest value of maximum acceleration of the former. It can also be observed that the values at the stable crushing zone, where the higher amount of energy is absorbed [54], are generally higher when applying the non-strain rate dependent, which is verified by calculating the mean force, defined as:

$$F_{mean} = \frac{\int F dl}{L} \quad (29)$$

where F is the force and L the total crushing distance. Additionally, the Specific Energy Absorption (SEA) [51] can be obtained:

$$SEA = \frac{F_{mean}}{\rho A} \quad (30)$$

where ρ is the density and A the cross sectional area. A summary of the crashworthiness metrics is shown in Table 8.

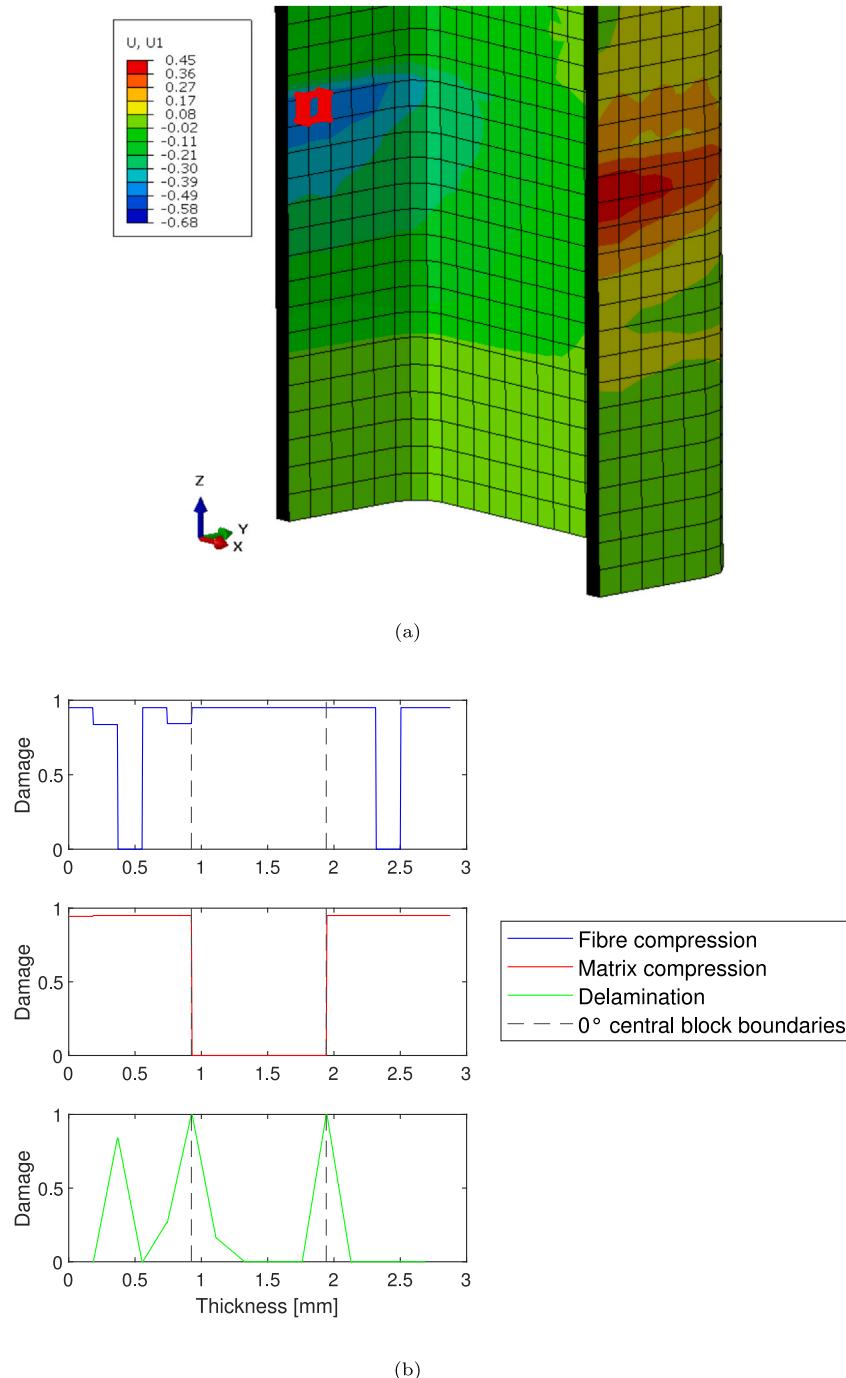


Fig. 15. (a) Element in the local buckling onset zone and (b) respective damage variables through its thickness ($t = 0.825$ ms after impact).

Table 8
Summary of crashworthiness metrics.

Criterion	F_{\max}	F_{mean}	SEA
Non-strain rate	90.39 kN	20.56 kN	37.69 kJ/kg
Strain rate	93.75 kN	15.70 kN	28.77 kJ/kg

As seen in Table 8, the non-strain rate dependent model leads to a higher mean force and consequently, a higher SEA, which was already expected from the kinetic energy evolution (Fig. 21(a)).

6. Conclusions

A three-dimensional, strain rate-dependent Hashin composite damage model is introduced and numerically implemented using a VUMAT subroutine to enhance the simulation of impact problems, with particular focus on composite fuselage components. This model is benchmarked against a well-documented low-velocity impact problem and demonstrates improved prediction accuracy compared to experimental results, reducing the absorbed energy value prediction error by at least 43% relatively to previous studies, primarily due to the incorporation

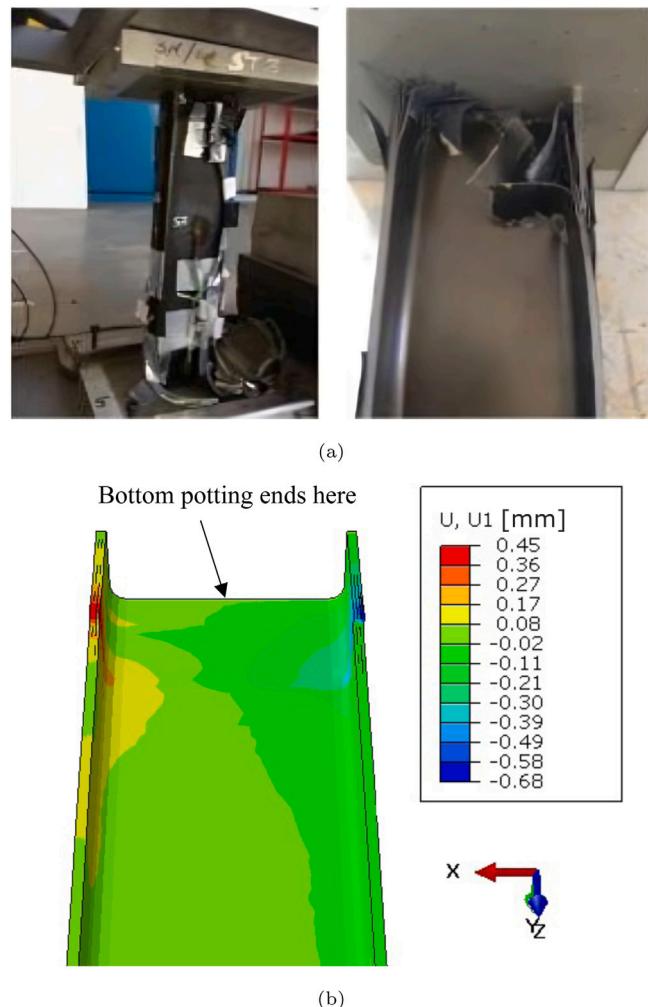


Fig. 16. (a) Stanchion at the end of experimental test [48] and (b) numerical results ($t = 0.825$ ms after impact).

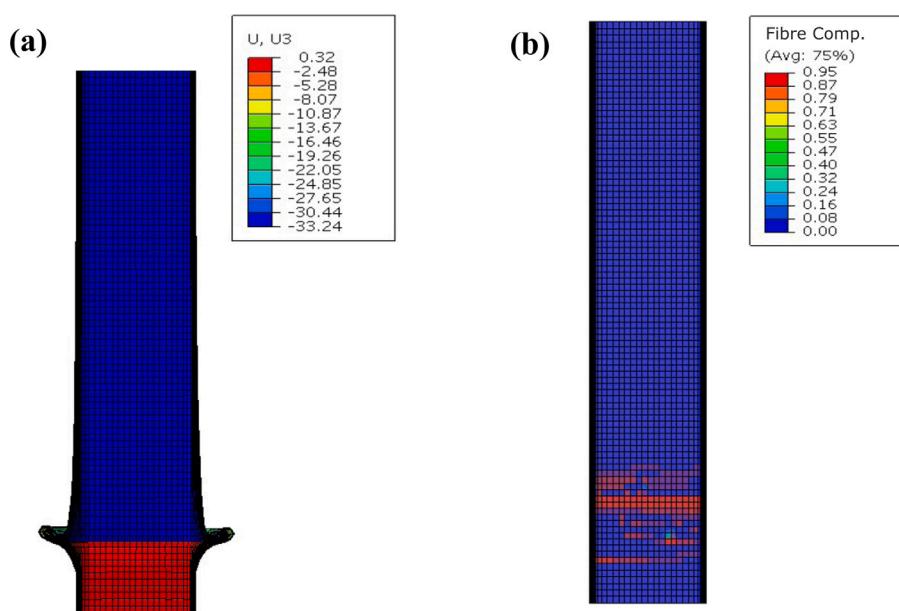


Fig. 17. Numerical (a) vertical displacement and (b) undeformed fibre compressive damage distribution with transparency.

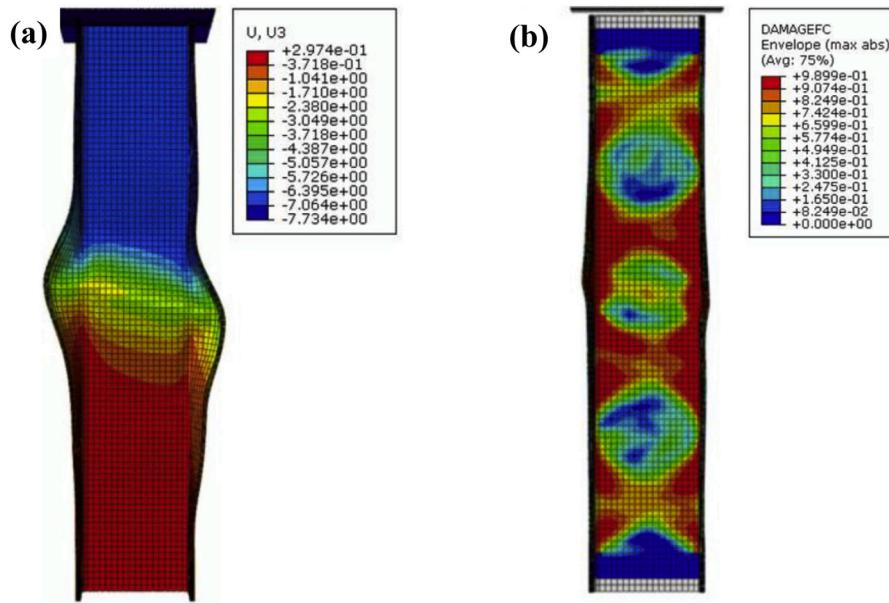


Fig. 18. Literature (a) vertical displacement at maximum compression and (b) fibre compressive damage distribution [47].

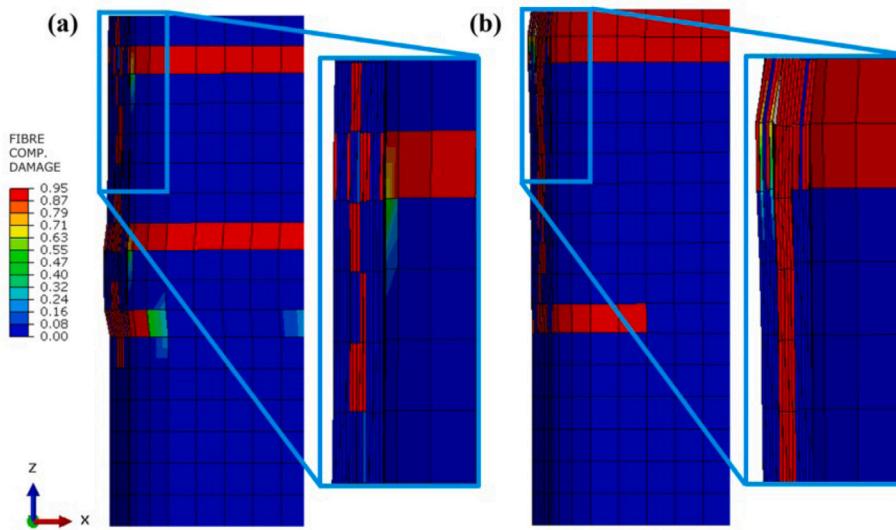


Fig. 19. (a) Non-strain rate and (b) strain rate dependent compressive fibre damage ($t = 0.825$ ms after impact).

of the strain rate effect. The damage model is also applied to simulate a high-energy impact on a composite stanchion of a fuselage structure documented in the literature. The numerical results show excellent agreement with experimentally observed damaged zones and failure modes. It leads to an increase in accuracy compared to previous studies and to the same numerical model employing a strain rate insensitive damage law, increasing the predicted value of peak force by nearly 4%. In conclusion, the introduced three-dimensional strain rate-dependent Hashin failure model proves to be highly effective for accurately predicting composite fuselage impact problems, indicating that incorporating strain rate effects could be beneficial for future simulations.

CRediT authorship contribution statement

Mário Miranda: Writing – review & editing, Writing – original draft, Visualization, Methodology, Investigation, Formal analysis. **Andrea Cini:** Writing – review & editing, Supervision, Resources, Conceptualization. **Antonio Raimondo:** Writing – review & editing, Supervision, Conceptualization. **Volnei Tita:** Writing – review & editing, Supervision, Conceptualization.

Declaration of competing interest

The authors declare that they have no known competing financial interests or personal relationships that could have appeared to influence the work reported in this paper.

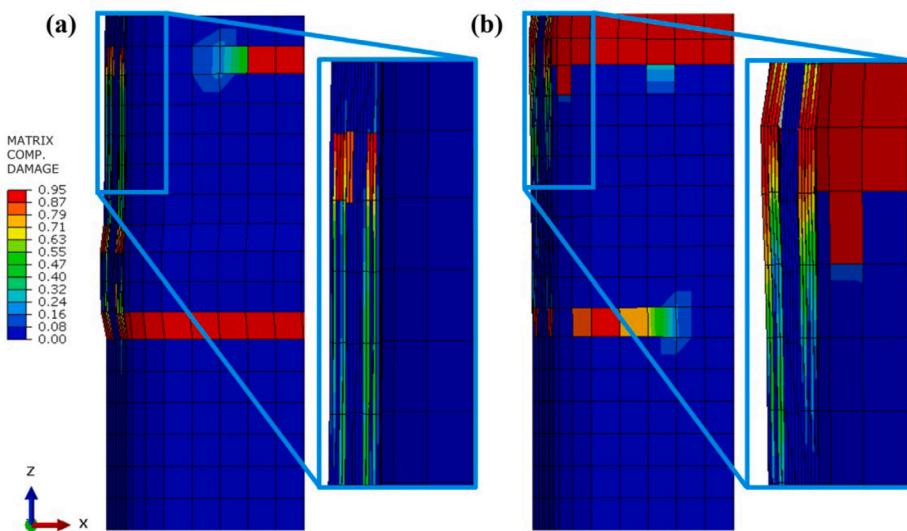
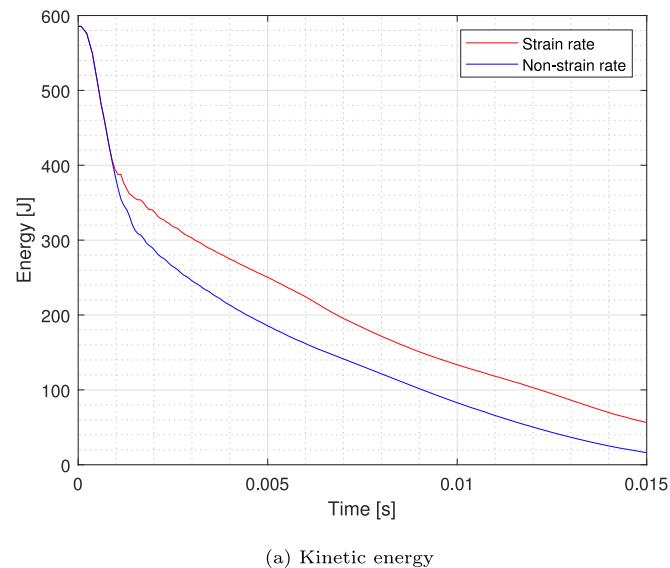
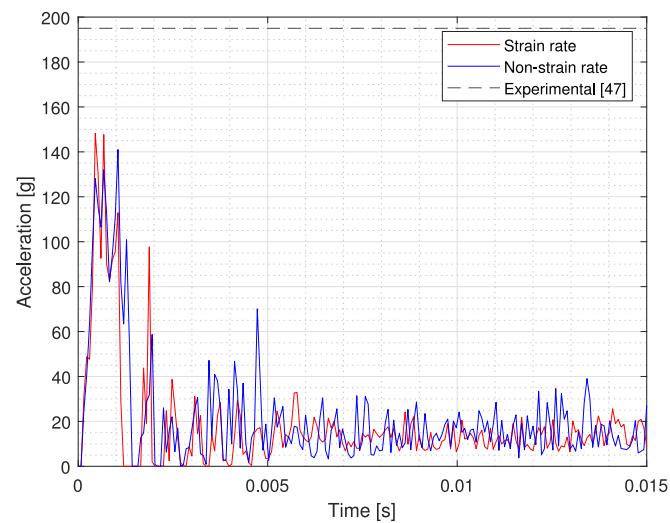


Fig. 20. (a) Non-strain rate and (b) strain rate dependent compressive matrix damage ($t = 0.825$ ms after impact).



(a) Kinetic energy



(b) Acceleration

Fig. 21. Evolution of impactor (a) kinetic energy and (b) acceleration employing both strain rate and non-strain rate dependent models.

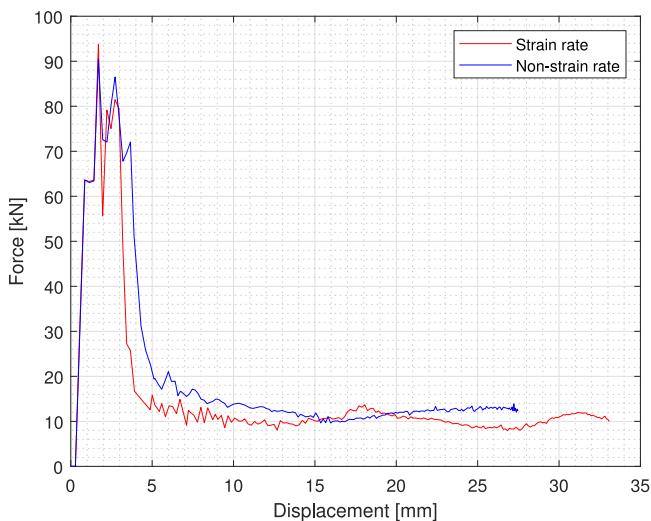


Fig. 22. Force-displacement curves employing both strain rate and non-strain rate dependent models.

Data availability

No data was used for the research described in the article.

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